

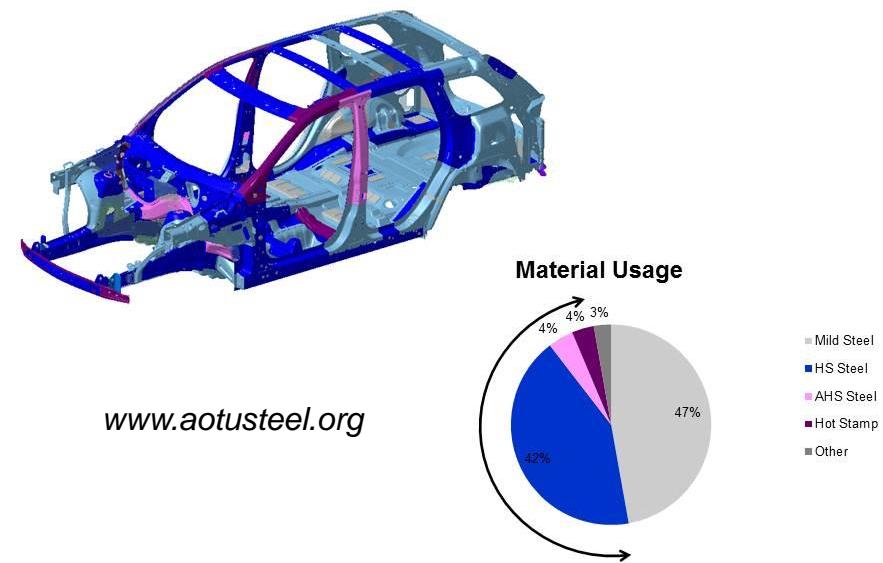
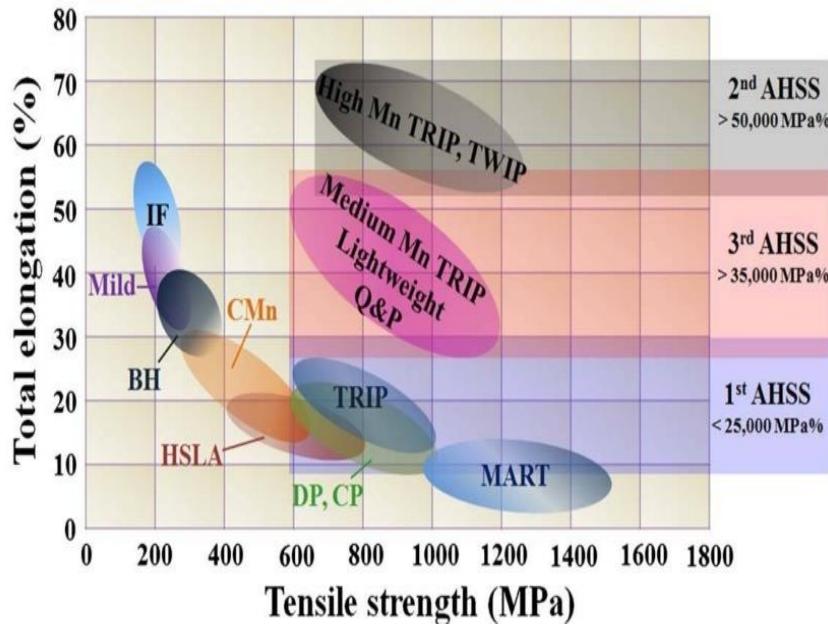
Kinetic of Martensite/austenite interface migration and carbon partitioning during the Q&P process

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School of Materials Science and Engineering,
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Introduction

- Vehicle emissions and exhaust
- Fuel consumption
- Safety and Lightweight

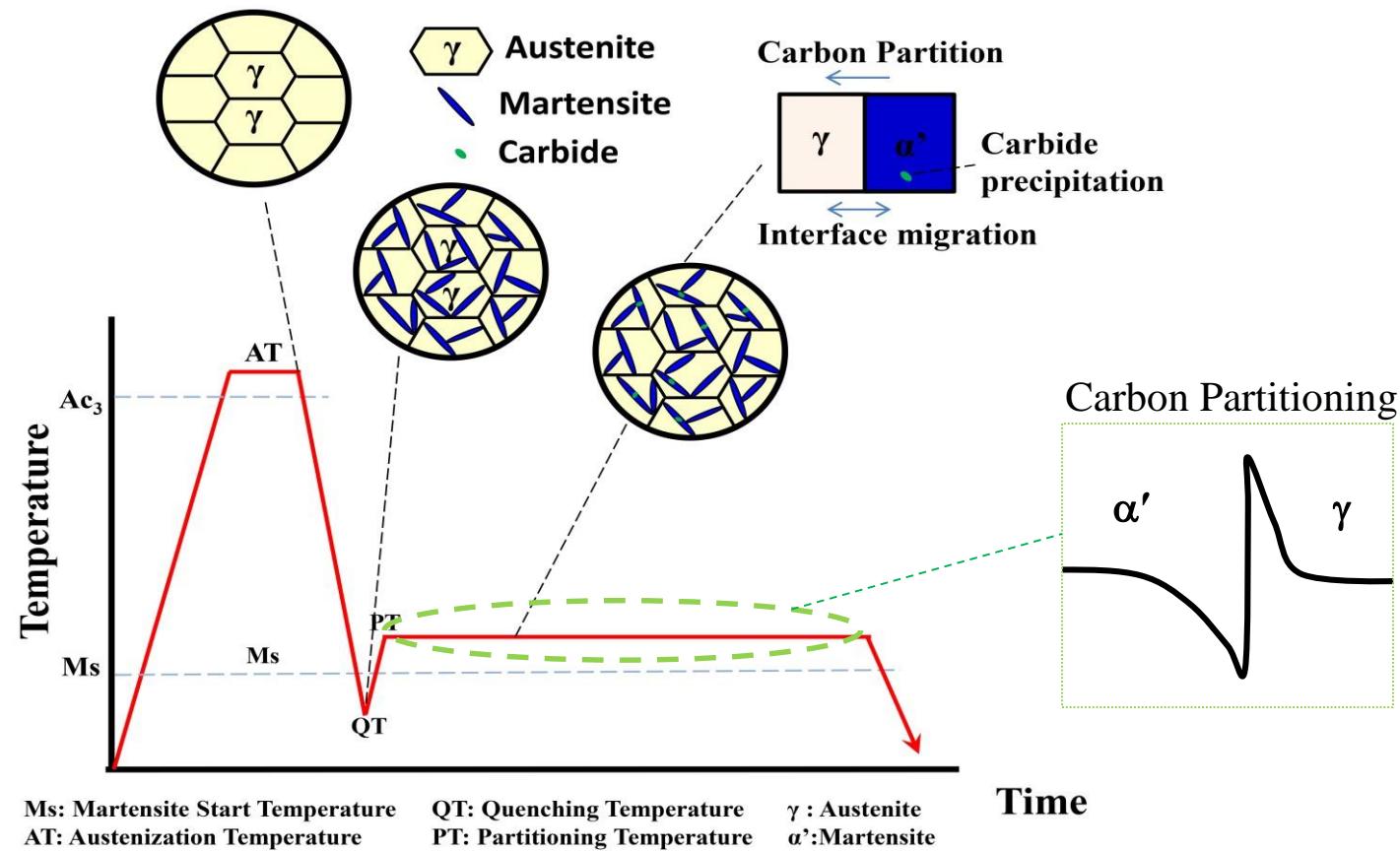


Retained Austenite

Retained Austenite + Martensite (Q&P)

*Retained Austenite + Ferrite + Bainitic Ferrite
Ferrite + Martensite*

Quenching & Partitioning Process



Scientific questions:

1. Is the α'/γ interface mobile or not? How?
2. Is there partitioning of substitutional alloying elements during the Q&P process?
3. Will partitioning of substitutional alloying elements affect the interface migration and carbon partitioning?

The Interface Migration or not?!

□ Immobile Interface : (*CCE Theory*)

- Toji et al, Acta Mater., 65 (2014) 215
- Seo et al, Acta Mater., 107 (2016) 354
- Seo et al, Acta Mater., 113 (2016) 124
- Bigg et al, JAC, 577S (2013) 695
- Thomas et al, MST, 30 (2014) 998

• • •

Dilatometer

Dilatometer

Dilatometer

In-situ Neutron diffractometry

EBSD

□ Mobile Interface :

- Zhong et al, JMST, 22 (2006) 751.

Ex-situ TEM

□ Interface migrates from BCC to FCC:

- Santofimia et al, MSEA, 527 (2010)
- Santofimia et al, Acta Mater., 59 (2011)
- Allain et al, Scripta Mater., 131 (2017)

• • •

Dilatometer

Dilatometer

In-situ HE-XRD

□ Interface migrates from FCC to BCC:

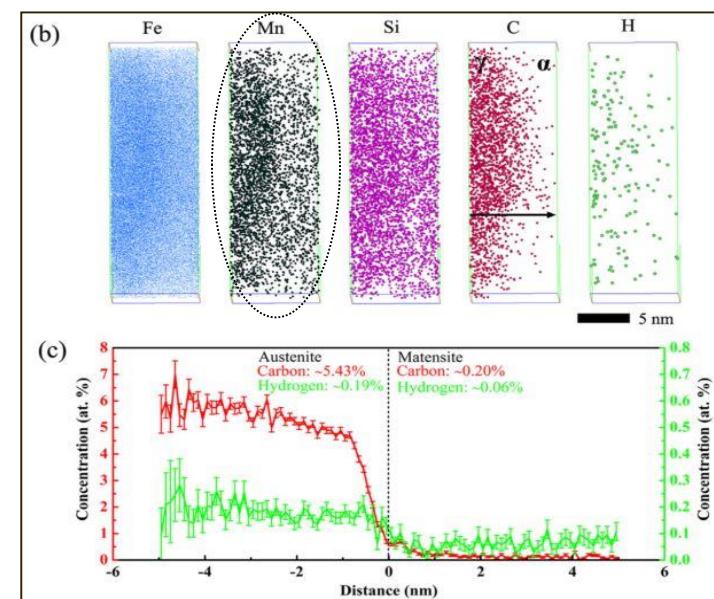
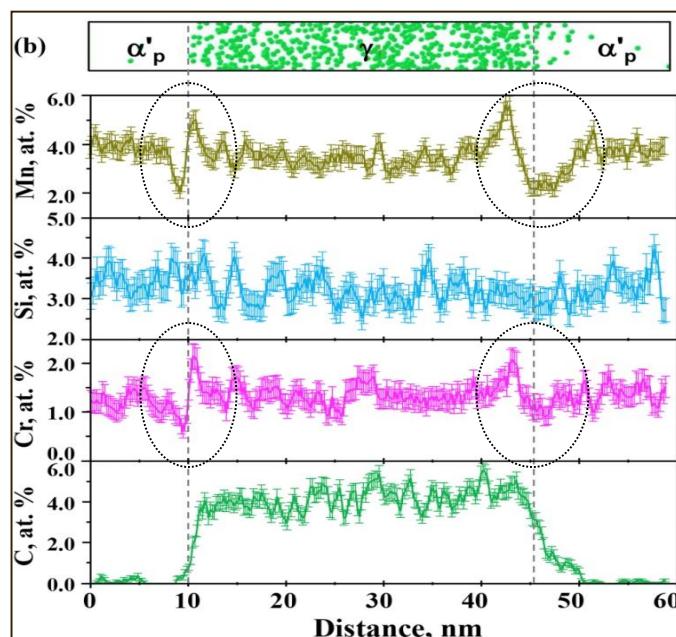
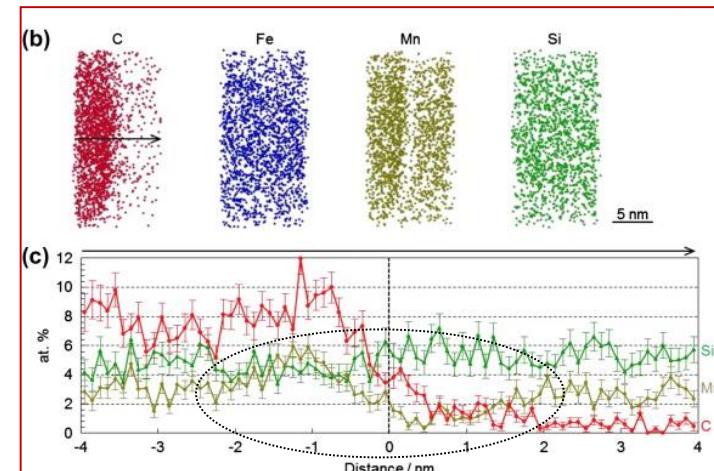
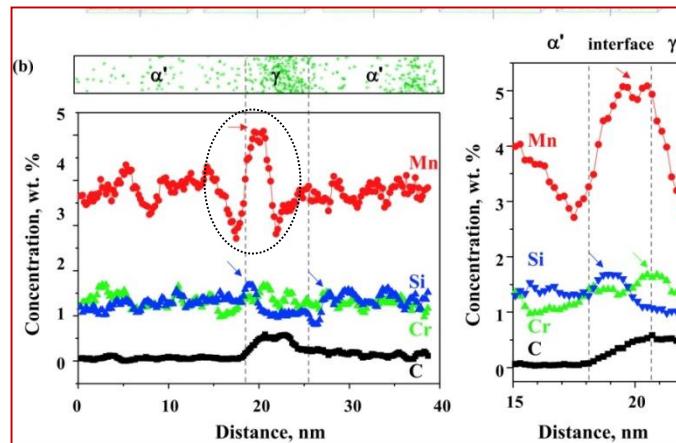
- De Knijf et al, Acta Mater., 90 (2015)
- Thomas et al, MST, 30 (2014) 998

In-situ HR-TEM

EBSD

Complex and Controversial !

Mn and Cr Partition or not?!

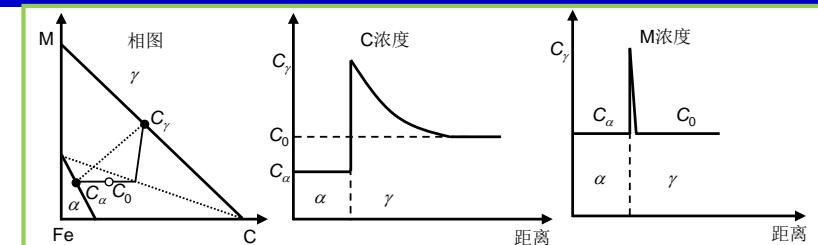


Models

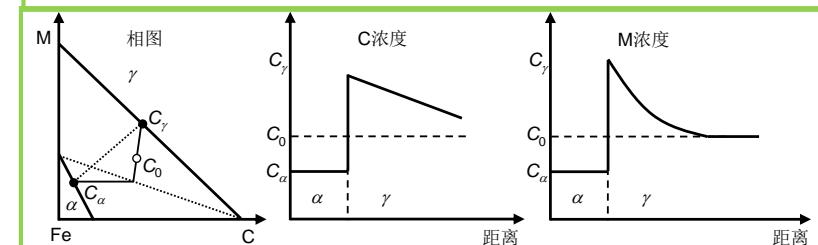
LE--Local equilibrium:

$$\mu_i^\gamma = \mu_i^\alpha$$

NP-LE :
C diffusion controlled



P-LE :
M diffusion controlled

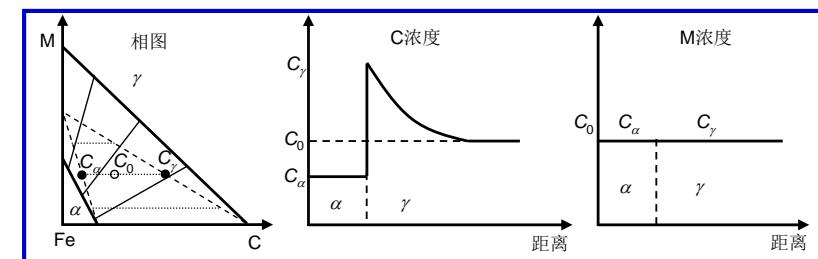


PE--Paraequilibrium:

$$\mu_C^\gamma = \mu_C^\alpha$$

$$(\mu_M^\gamma - \mu_M^\alpha) = -\frac{X_{Fe}}{X_M} (\mu_{Fe}^\gamma - \mu_{Fe}^\alpha)$$

PE :
C diffusion
Mn indiffusible



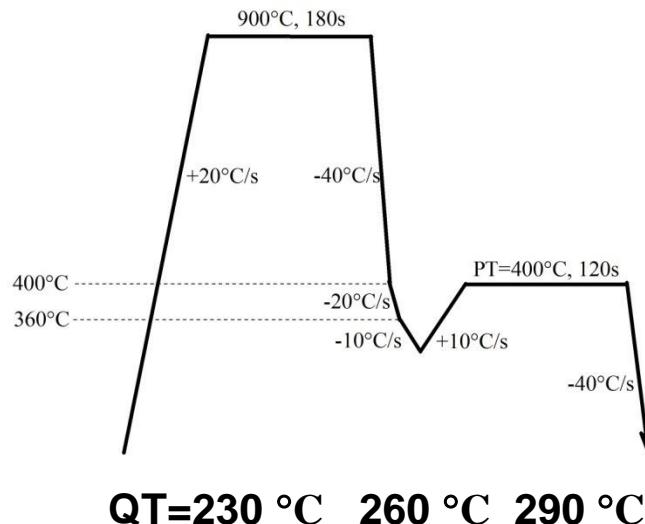
CCE—Constrain Carbon Equilibrium:

$$\mu_C^\gamma = \mu_C^\alpha$$

Martensite/austenite interface is immobile

Experiments

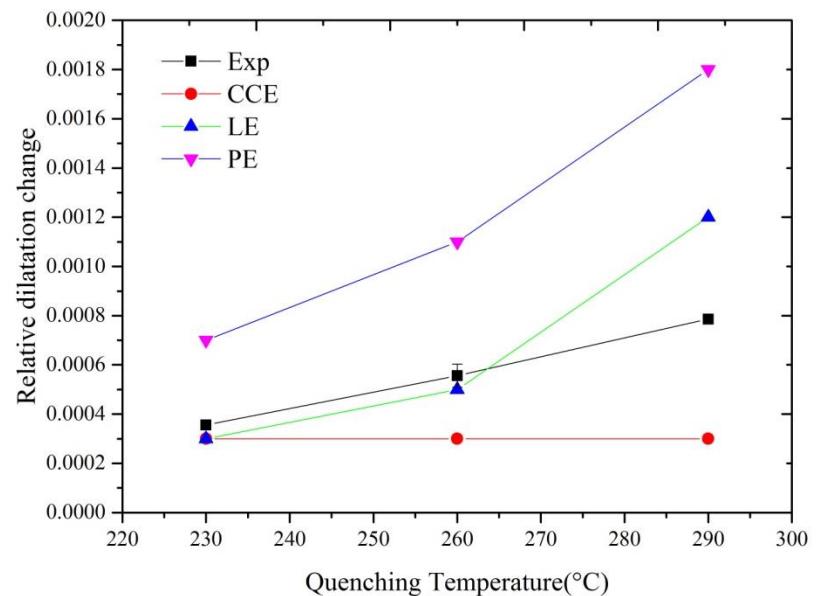
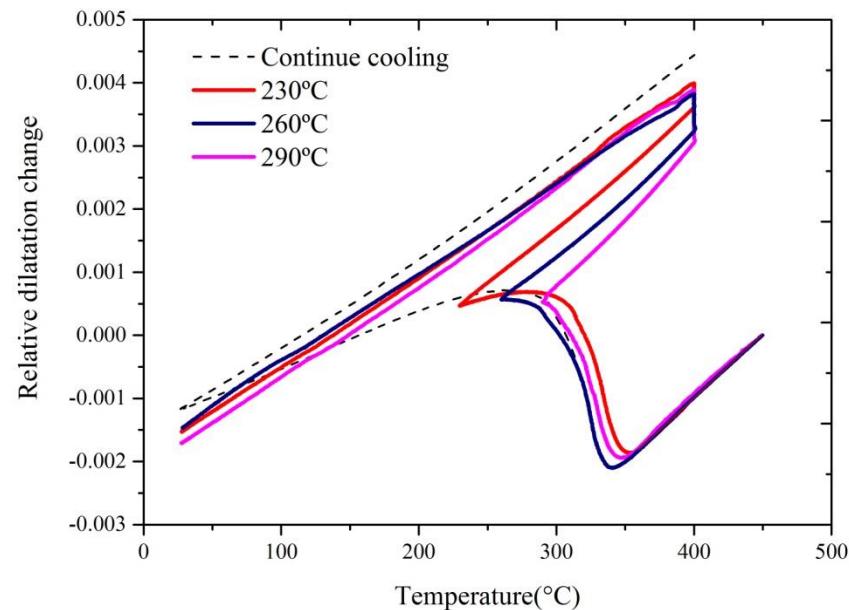
Fe-0.25C-2.1Mn-1.1Si (wt.%)



Calculation of relative dilatation strain:

$$\frac{\Delta L}{L_0} = \frac{L - L_0}{L_0} \approx \frac{1}{3} \frac{V - V_0}{V_0}$$

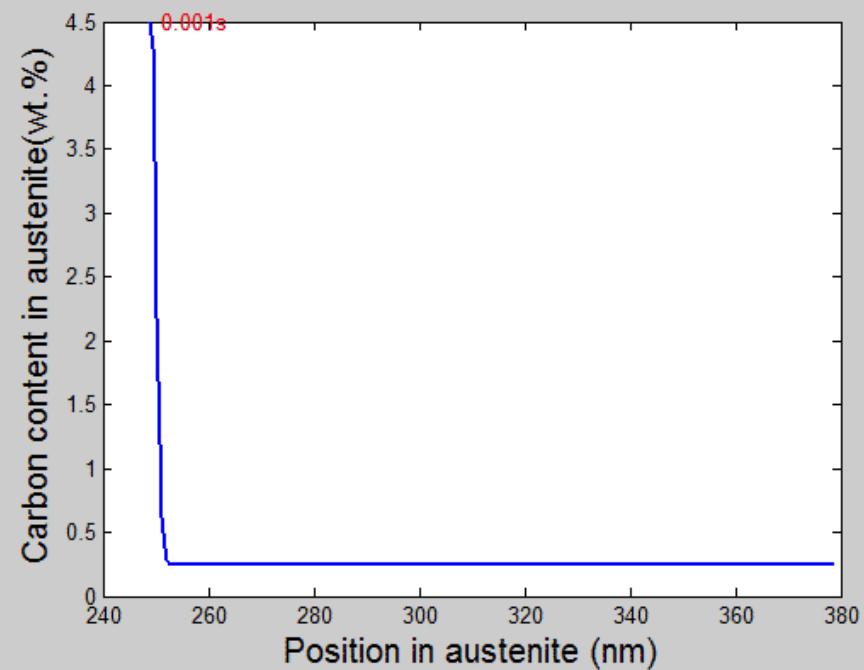
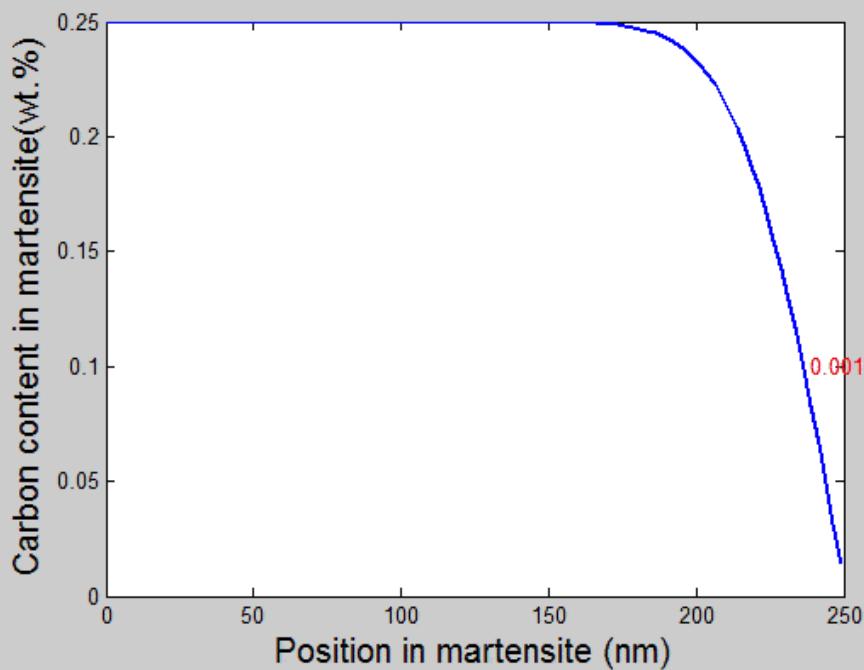
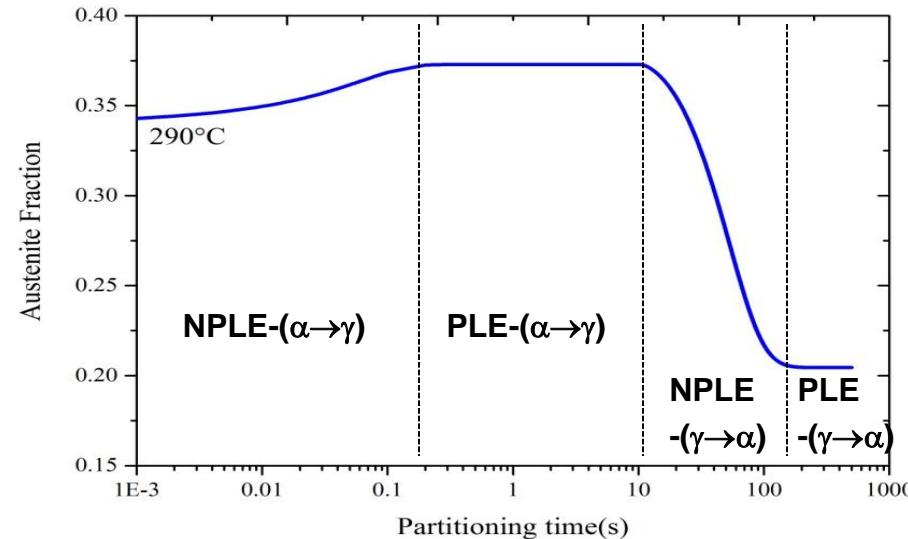
$$\frac{\Delta L}{L_0} = \frac{2V_{\alpha'}(1-x_C^{\alpha'})a_{\alpha'}^2c_{\alpha'} + [1-V_{\alpha'}-(x_{C0}^{\gamma}-V_{\alpha'}x_C^{\alpha'})]a_{\gamma}^3 - (1-x_{C0}^{\gamma})a_{\gamma0}^3}{3(1-x_{C0}^{\gamma})a_{\gamma0}^3}$$



Case for martensite growth

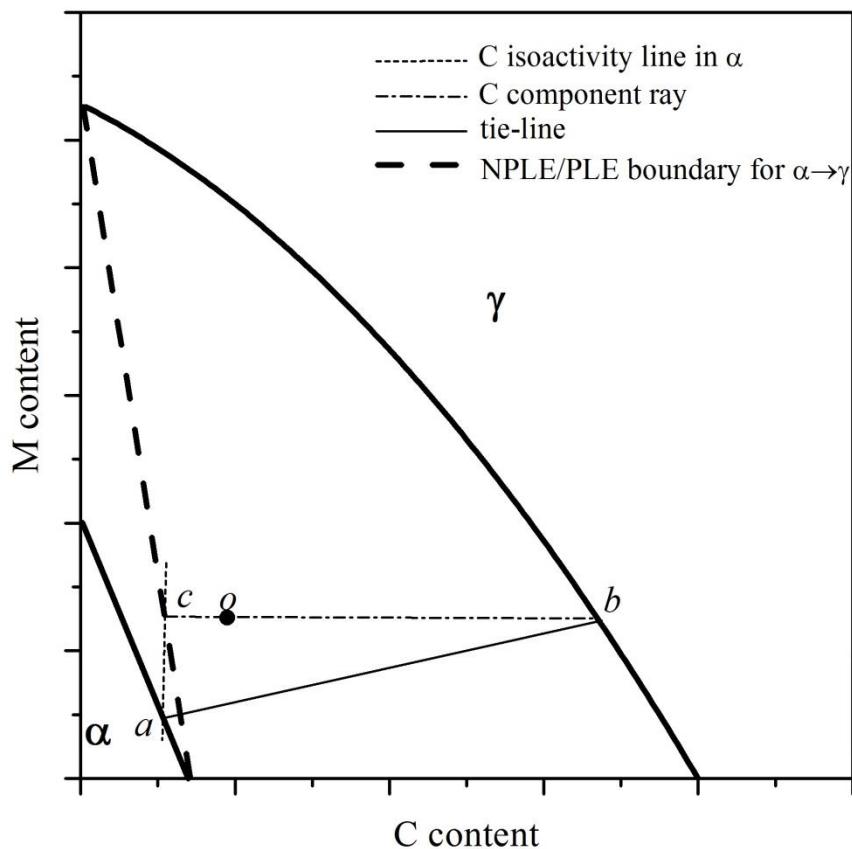
QT=290°C

PT=400°C



Case for martensite growth

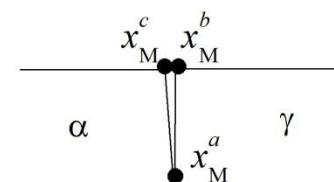
Stage I: NPLE-($\alpha \rightarrow \gamma$)



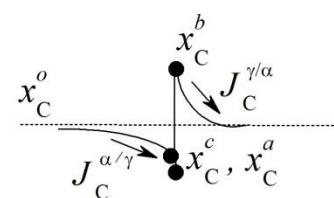
a)

Controlled by Carbon diffusion in martensite

NPLE Mode:



M Profile



C Profile

$$v = \frac{J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha} - x_C^{\alpha/\gamma}}$$

$$J_i^\phi = -\frac{D_i^\phi x_i^\phi}{RT} \nabla \mu_i^\phi$$

$$x_C^{\alpha'} = x_C^\gamma = x_{C0}$$

$$\nabla \mu_C^\alpha > \nabla \mu_C^\gamma$$

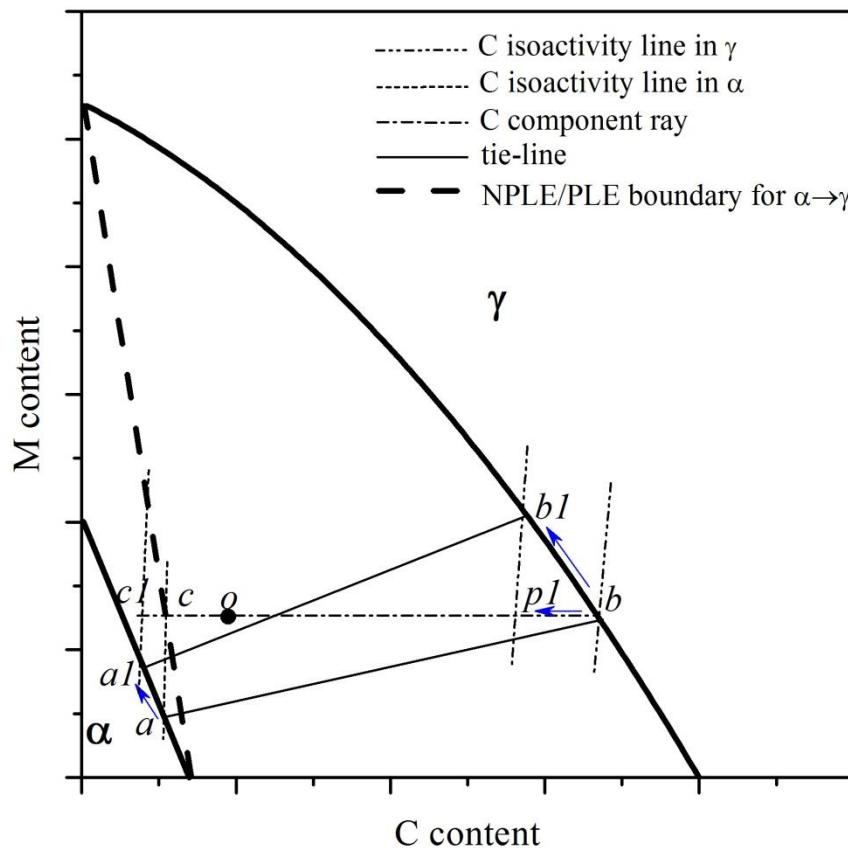
$$D_C^\alpha \square D_C^\gamma$$

$$J_C^{\alpha/\gamma} \square J_C^{\gamma/\alpha}$$

$$v = \frac{J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha} - x_C^{\alpha/\gamma}} \approx -\frac{J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha}}$$

Case for martensite growth

Stage II: PLE-($\alpha \rightarrow \gamma$)

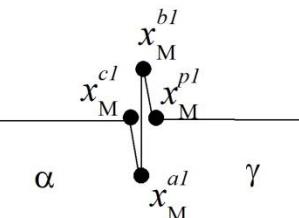


b)

$$\nu \approx 0 \quad \longrightarrow \quad D_{\text{Mn}}^{\alpha} \square \quad D_{\text{Mn}}^{\gamma}$$

Mn spike in martensite decrease : aa1 line
Mn spike in austenite is built up : bb1 line

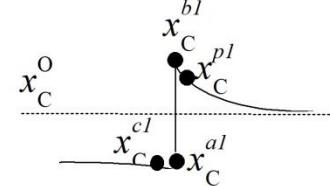
PLE Mode:



$$J_C^{\alpha/\gamma} \downarrow$$

$$J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma} \approx 0$$

M Profile

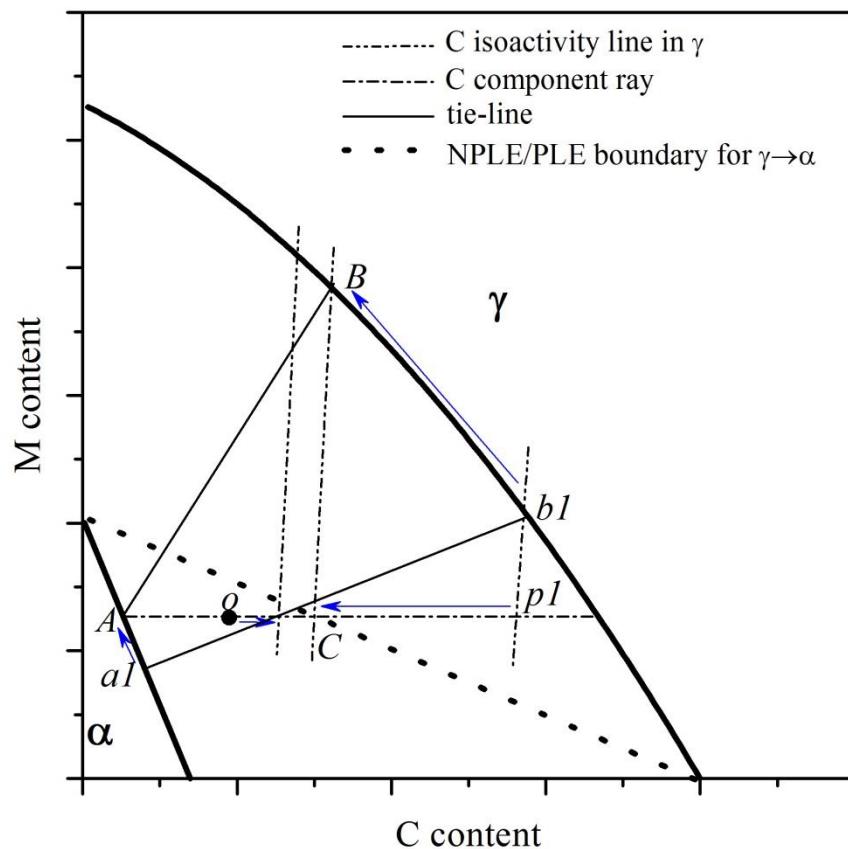


$$\nu = \frac{J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha} - x_C^{\alpha/\gamma}} \approx 0$$

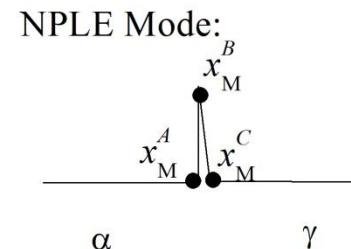
C Profile

Case for martensite growth

Stage III: NPLE-($\gamma \rightarrow \alpha$)

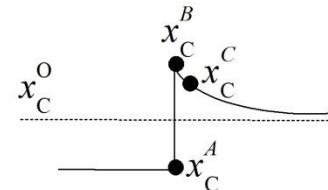


c)



M Profile

$$v = \frac{J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha} - x_C^{\alpha/\gamma}} \approx \frac{J_C^{\gamma/\alpha}}{x_C^{\gamma/\alpha}}$$

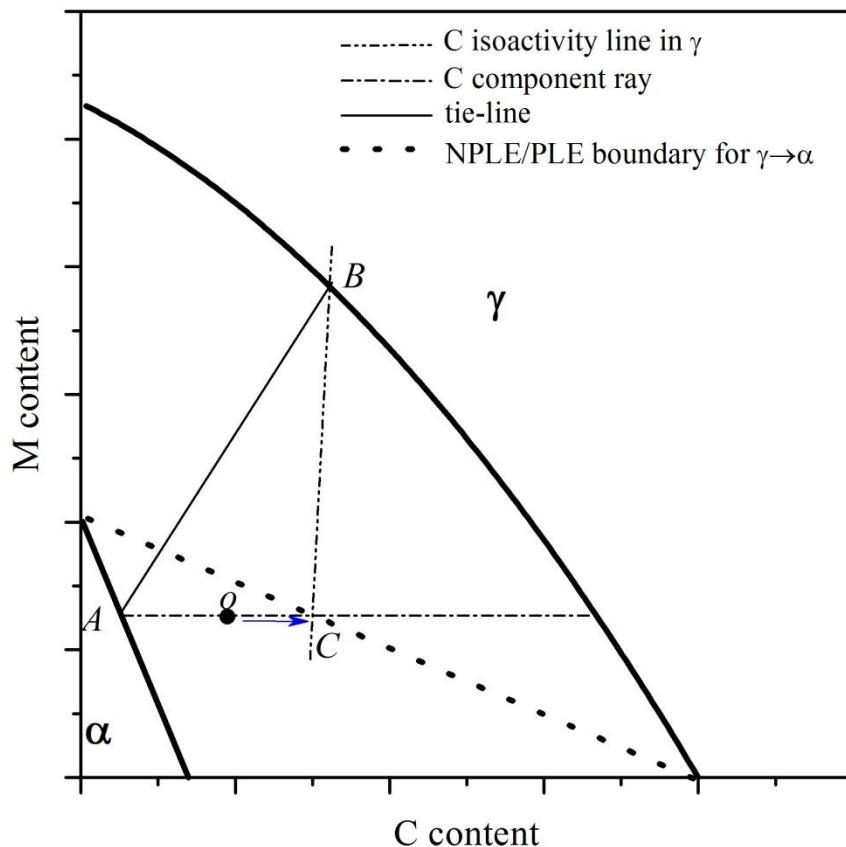


C Profile

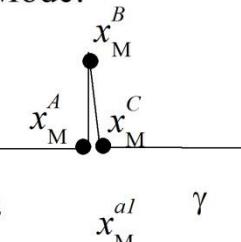
Controlled by Carbon diffusion in austenite

Case for martensite growth

Stage IV: PLE-($\gamma \rightarrow \alpha$)



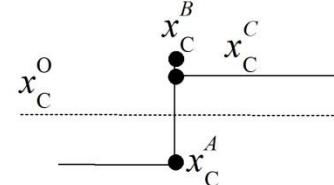
PLE Mode:



$$J_C^{\gamma/\alpha} \downarrow$$

$$J_C^{\gamma/\alpha} \approx 0$$

M Profile



$$\nu = \frac{J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha} - x_C^{\alpha/\gamma}} \approx 0$$

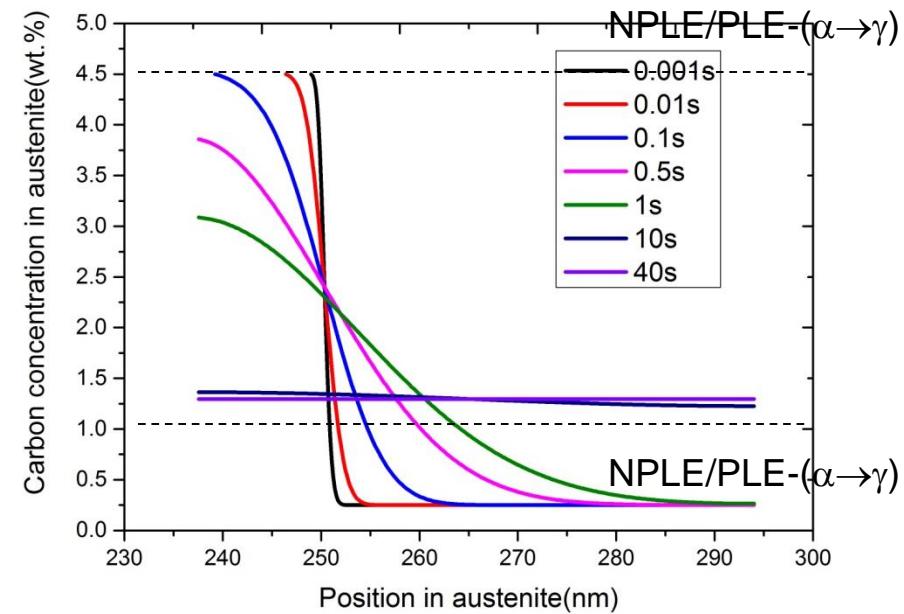
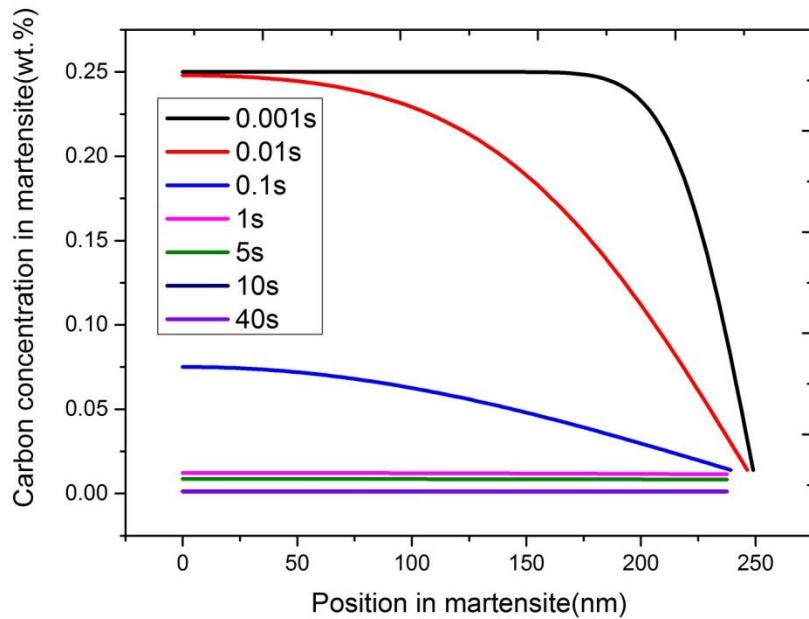
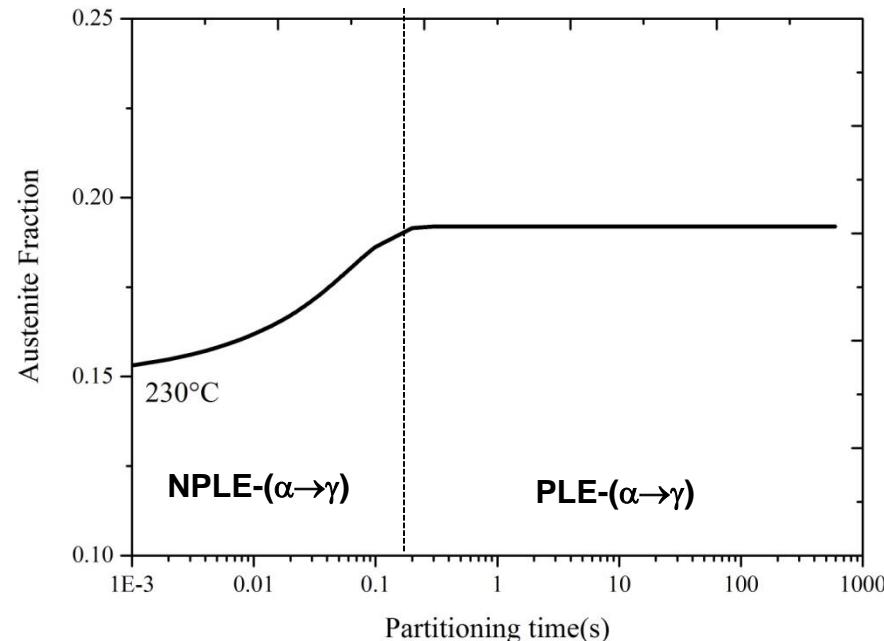
C Profile

d)

Carbon concentration in austenite will be determined by NPLE/PLE-($\gamma \rightarrow \alpha$) (i.e. Point C)

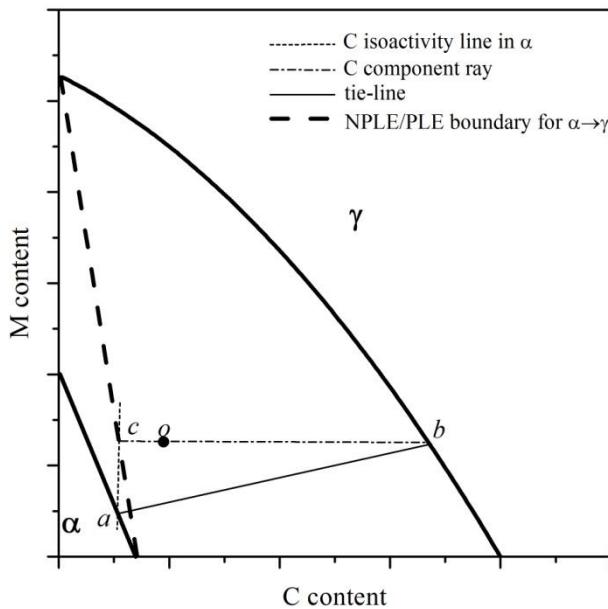
Case for Immobile interface

QT=230°C



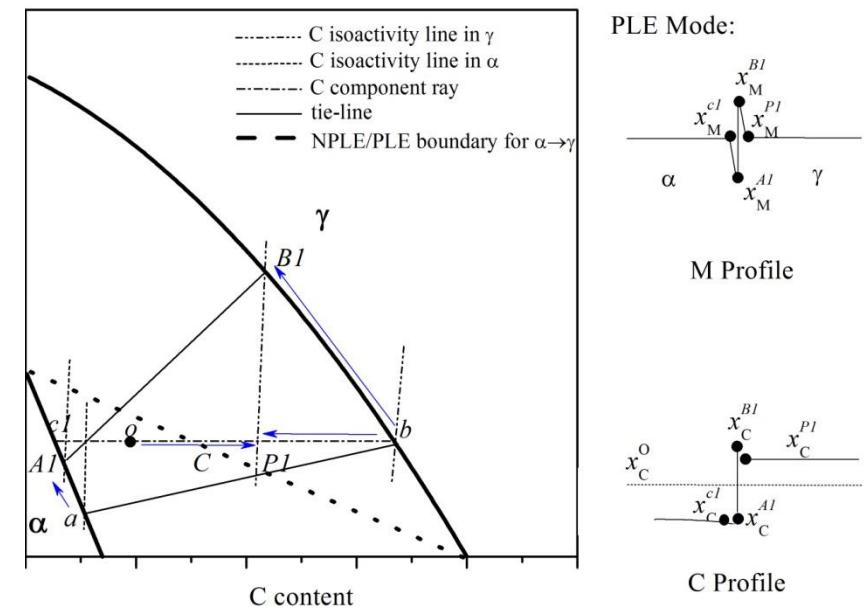
Case for Immobile interface

Stage I: NPLE-($\alpha \rightarrow \gamma$)



a)

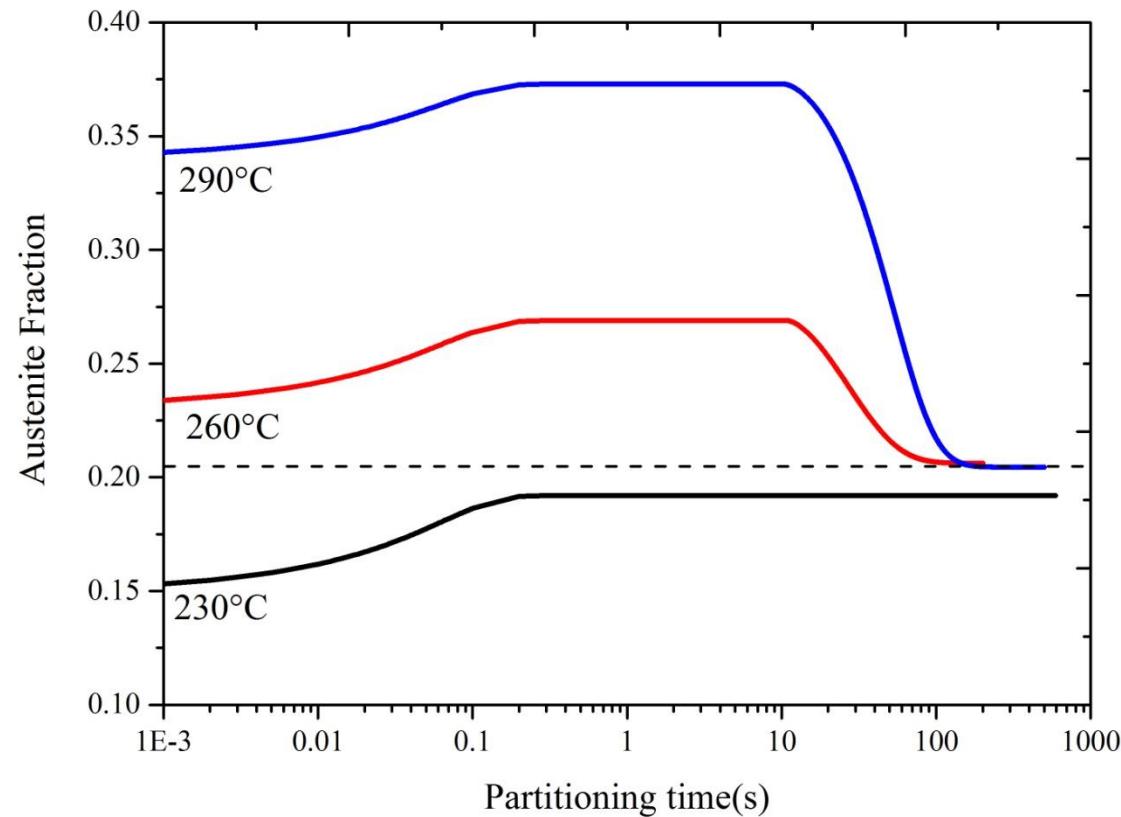
Stage II: PLE-($\alpha \rightarrow \gamma$)



b)

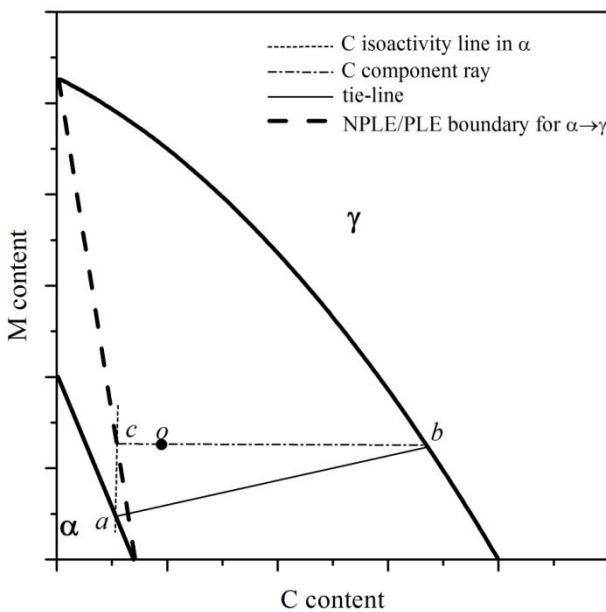
Carbon concentration in austenite (*Point P1*) will be located between NPLE/PLE-($\alpha \rightarrow \gamma$) (i.e. *Point b*) and NPLE/PLE-($\gamma \rightarrow \alpha$) (i.e. *Point C*)

Effect of QT on Interface migration

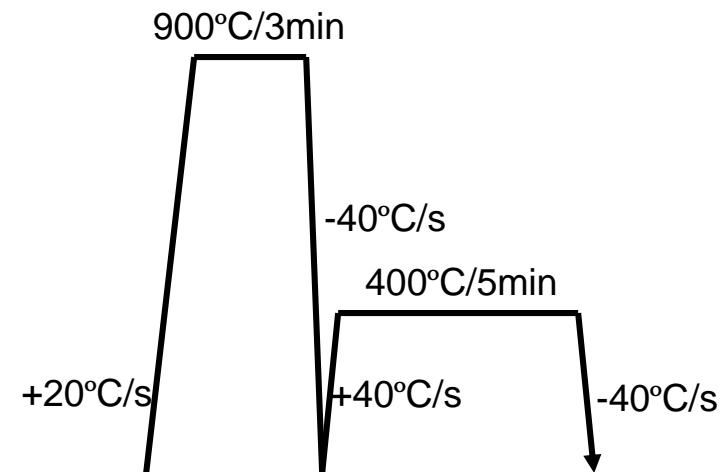


Extreme cases for austenite growth

Stage I: NPLE-($\alpha \rightarrow \gamma$)



Fe-0.59C-2.9Mn-2.0Si (wt.%)



QT=Room temperature

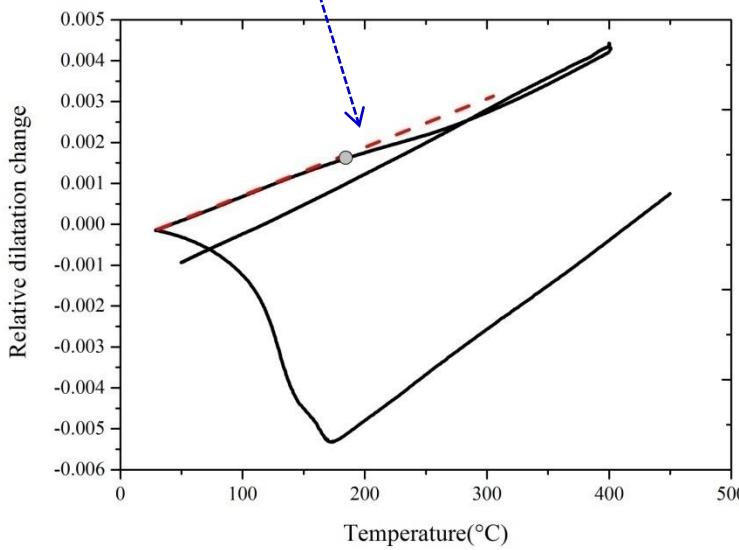
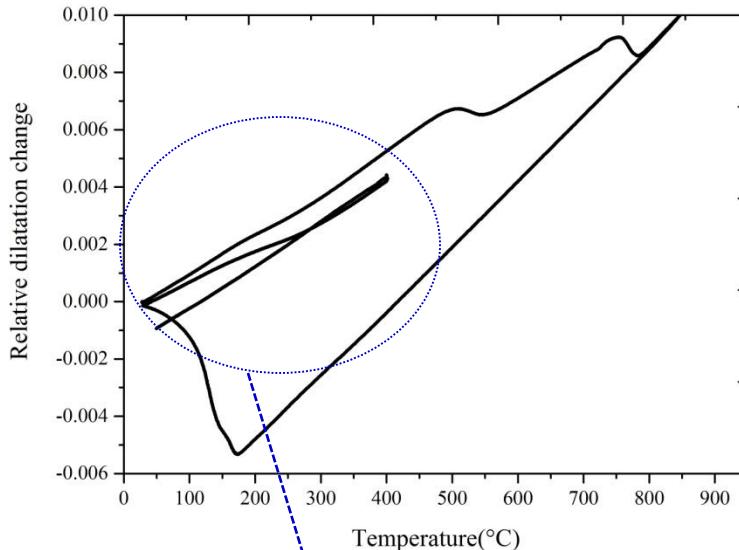
$V\gamma \approx 8\% \text{ at QT}$

$$\nu = \frac{J_C^{\gamma/\alpha} - J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha} - x_C^{\alpha/\gamma}} \approx -\frac{J_C^{\alpha/\gamma}}{x_C^{\gamma/\alpha}}$$

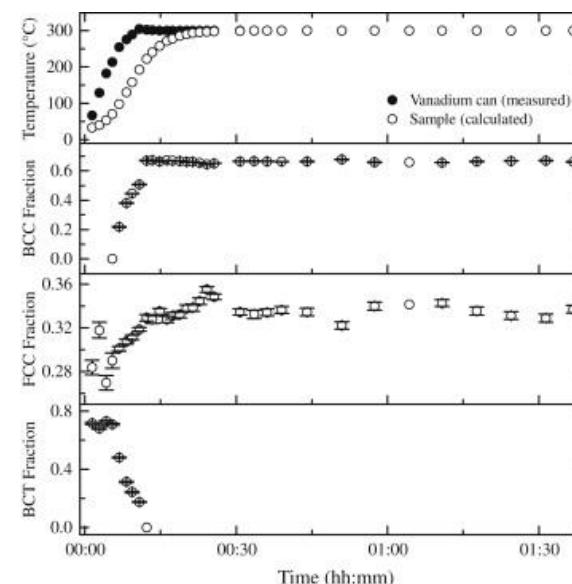
Controlled by Carbon diffusion in martensite

Extreme cases for austenite growth

Fe-0.59C-2.9Mn-2.0Si (wt.%)

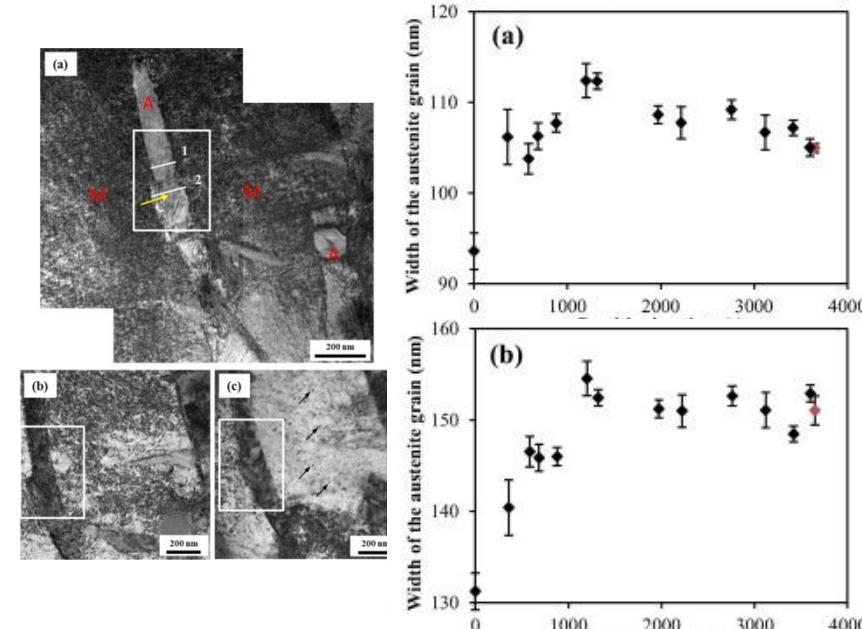


Fe-0.64C-4.57Mn-1.3Si (wt.%)

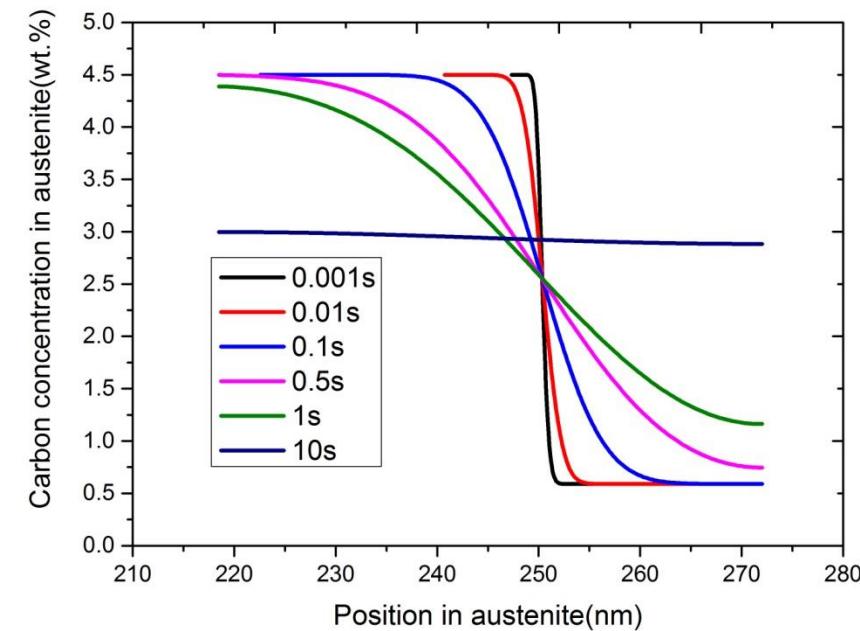
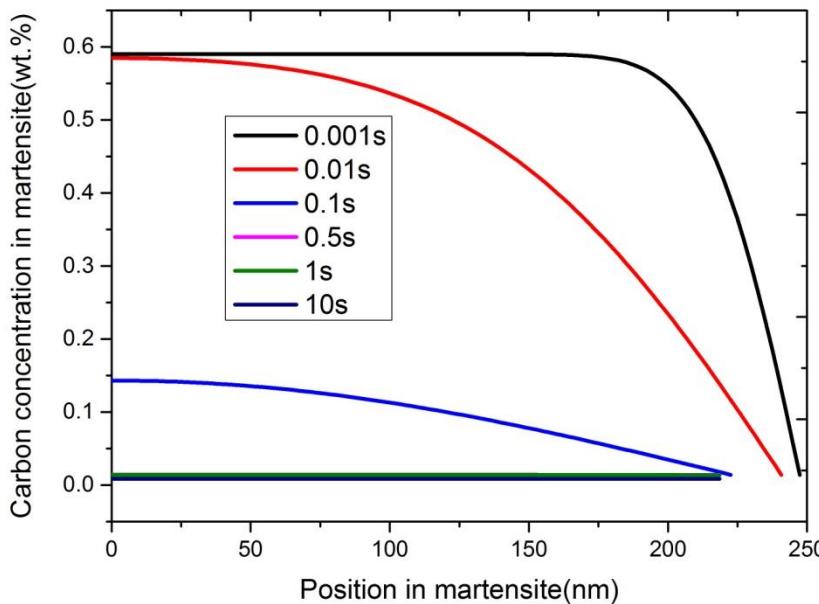
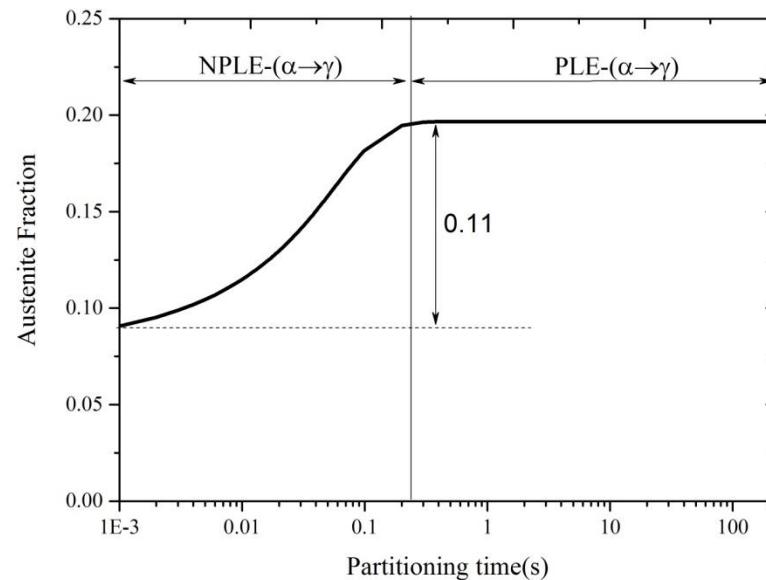


Bigg et al.
JAC
2013

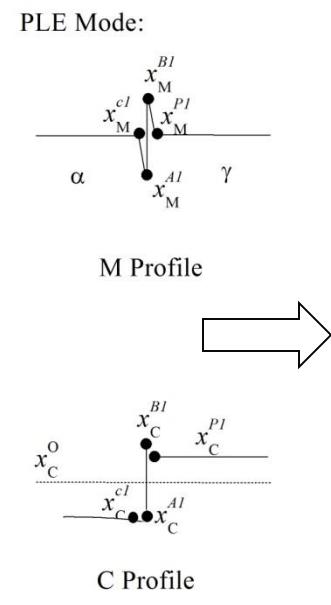
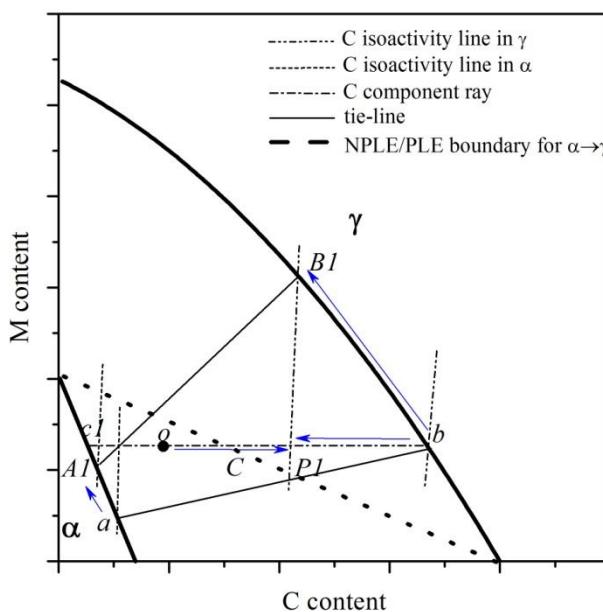
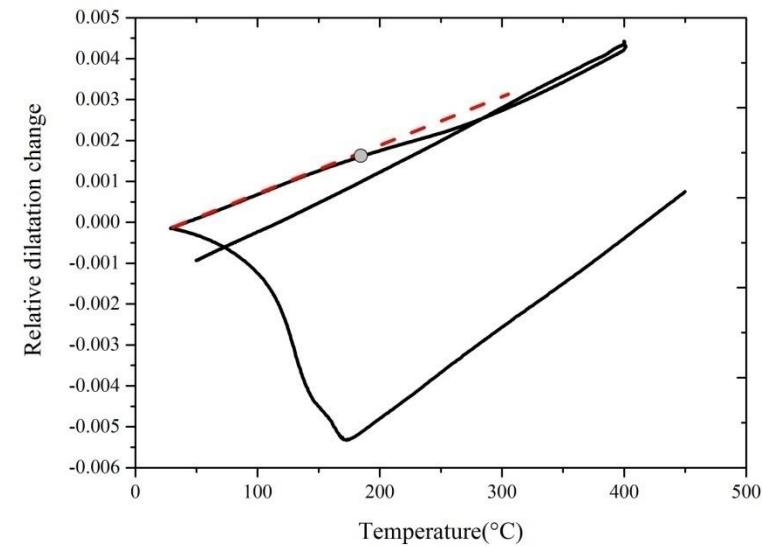
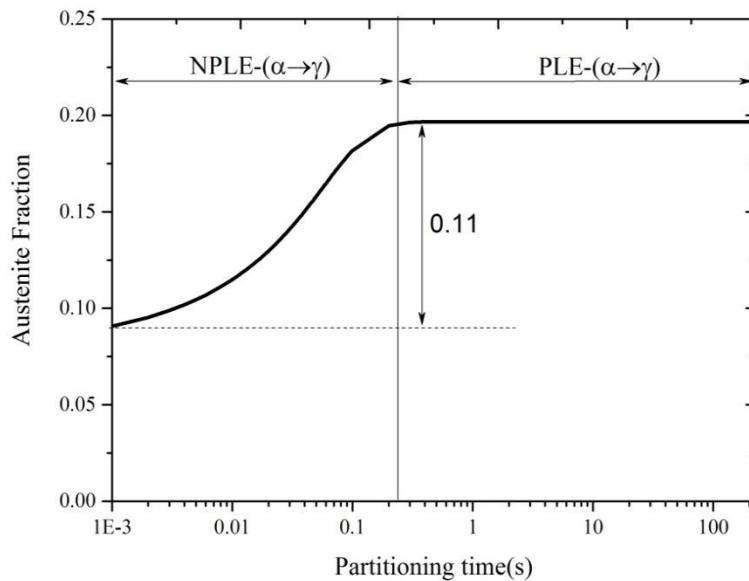
Fe-1.0C-3.0Mn-1.5Si (wt.%)



Extreme cases for austenite growth



Extreme cases for austenite growth



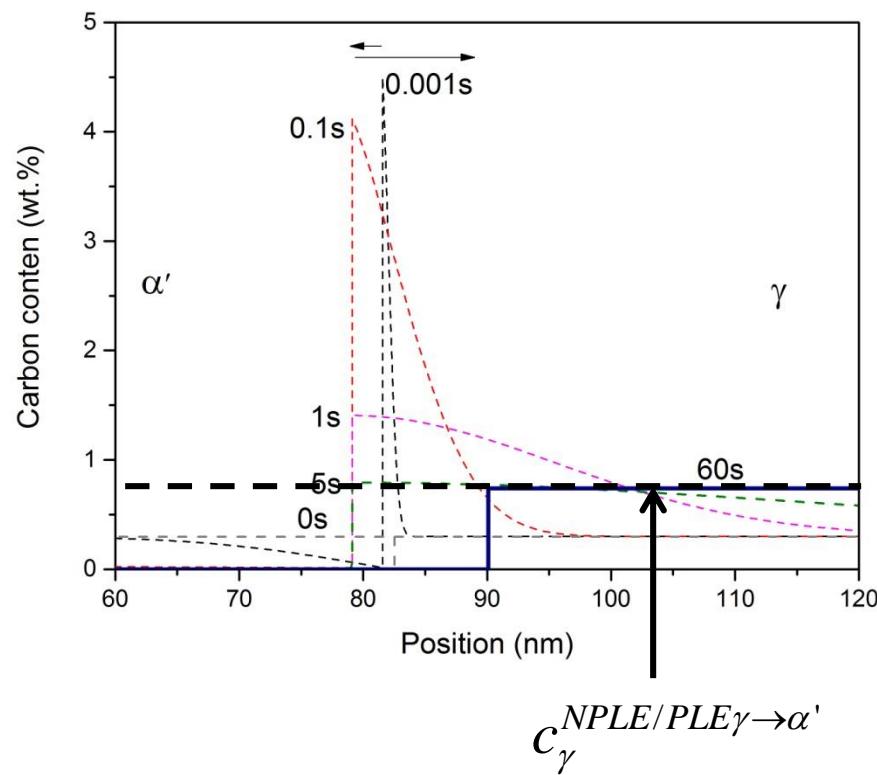
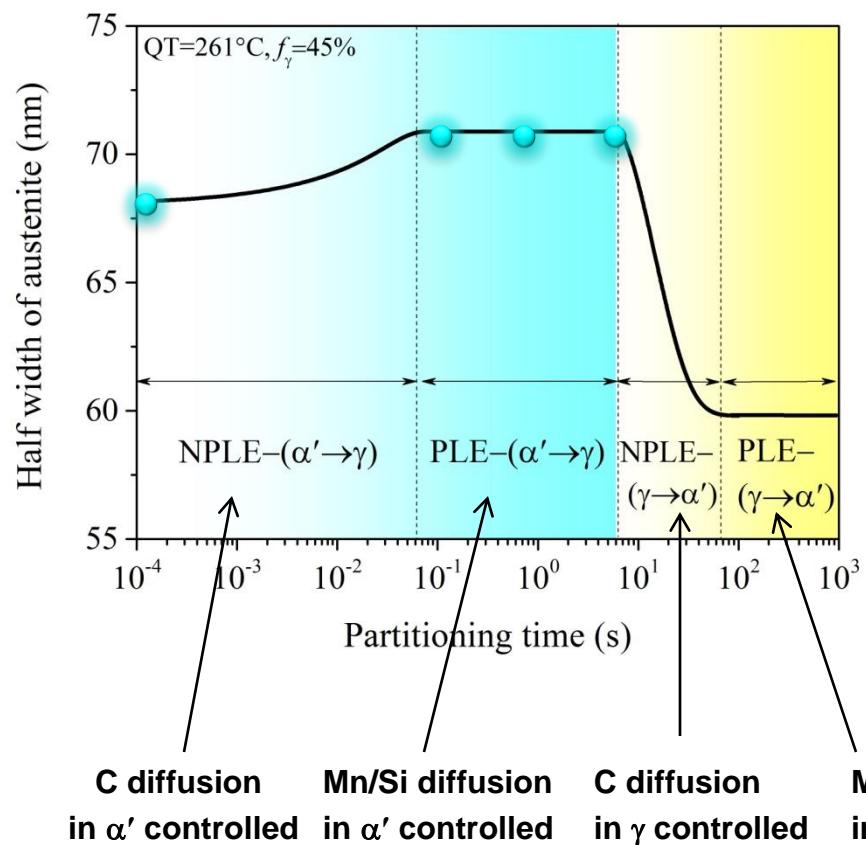
Conclusion

- Interfacial partitioning of substitutional alloying elements plays a significant role in interface migration behavior and C content in austenite.
- Kinetics of interface migration. Based on the LE model, the kinetics of interface migration can be generally divided into four stages: NPLE-($\alpha' \rightarrow \gamma$) → PLE-($\alpha' \rightarrow \gamma$) → NPLE-($\gamma \rightarrow \alpha'$) → PLE-($\gamma \rightarrow \alpha'$). Presence of the $\gamma \rightarrow \alpha'$ stages or not depends on QT.
- Alloying elements partitioning. After the partitioning process, C content in austenite could be located between the NPLE/PLE boundaries for the $\gamma \rightarrow \alpha'$ and $\alpha' \rightarrow \gamma$ transformation. Substitutional alloying elements partitioning across the interface is predicted to occur.

Thank You !

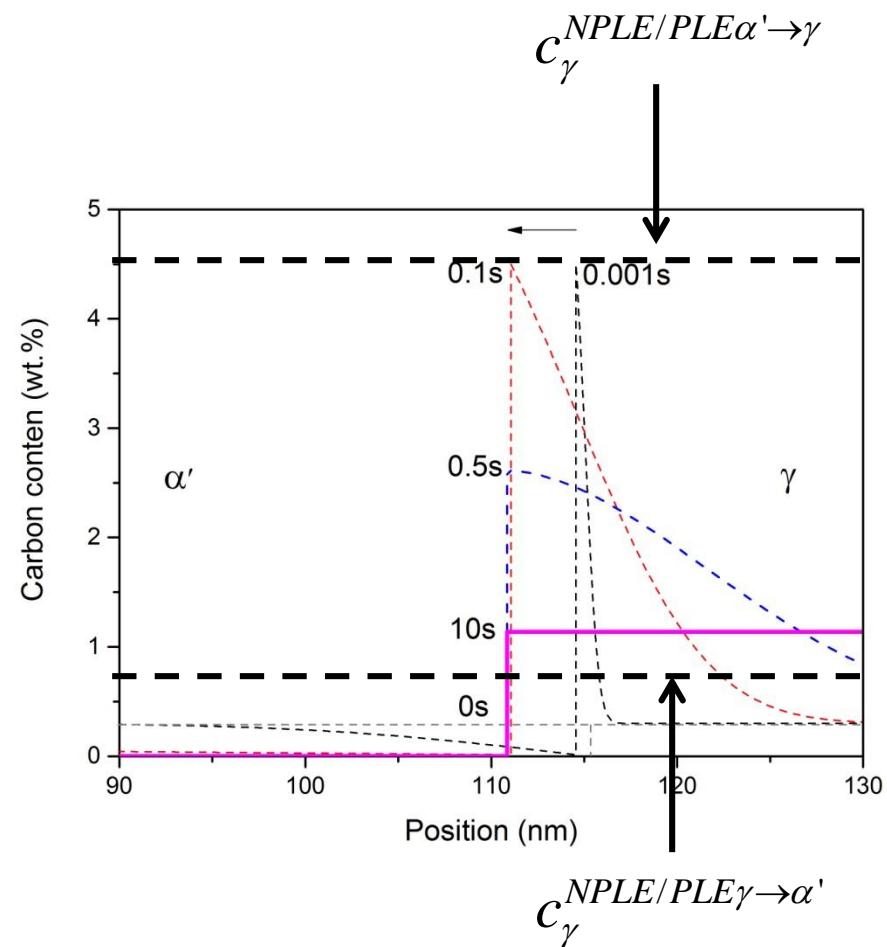
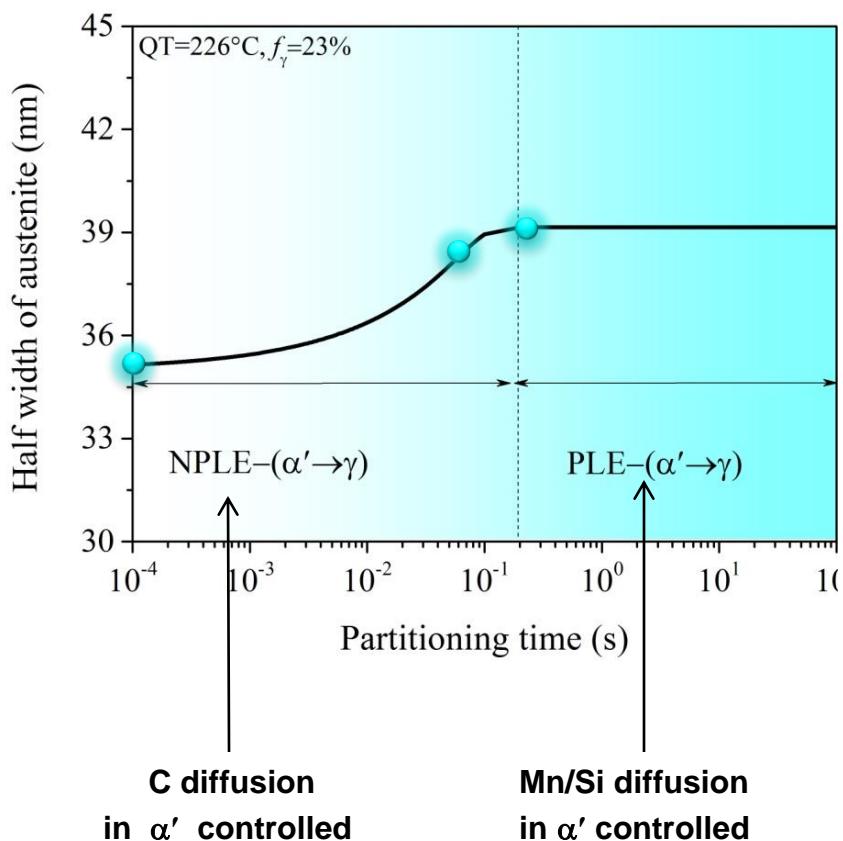
Simulations by the QP-LE Model

QT=261°C ($f_\gamma=45\%$), PT=400 °C



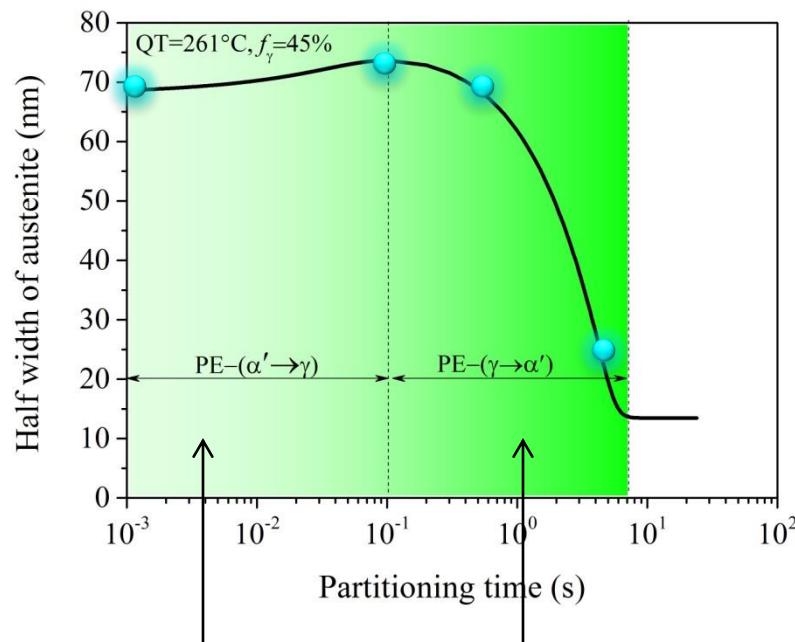
Simulations by the QP-LE Model

QT=226°C ($f_\gamma=23\%$), PT=400 °C



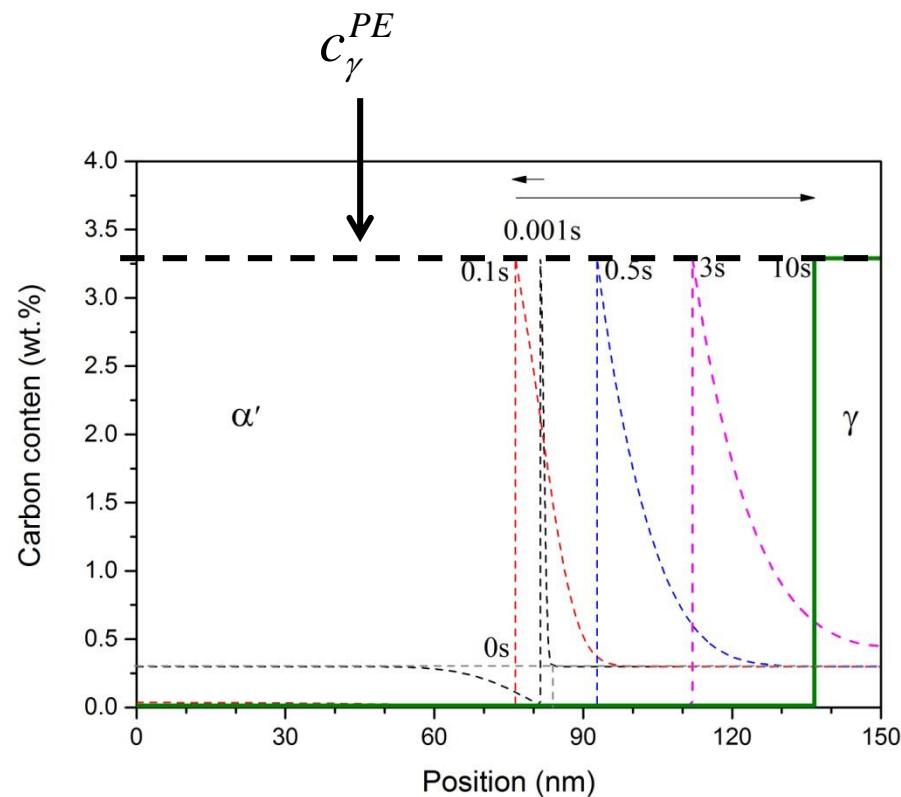
Simulations by the QP-PE Model

QT=261 °C, PT=400 °C



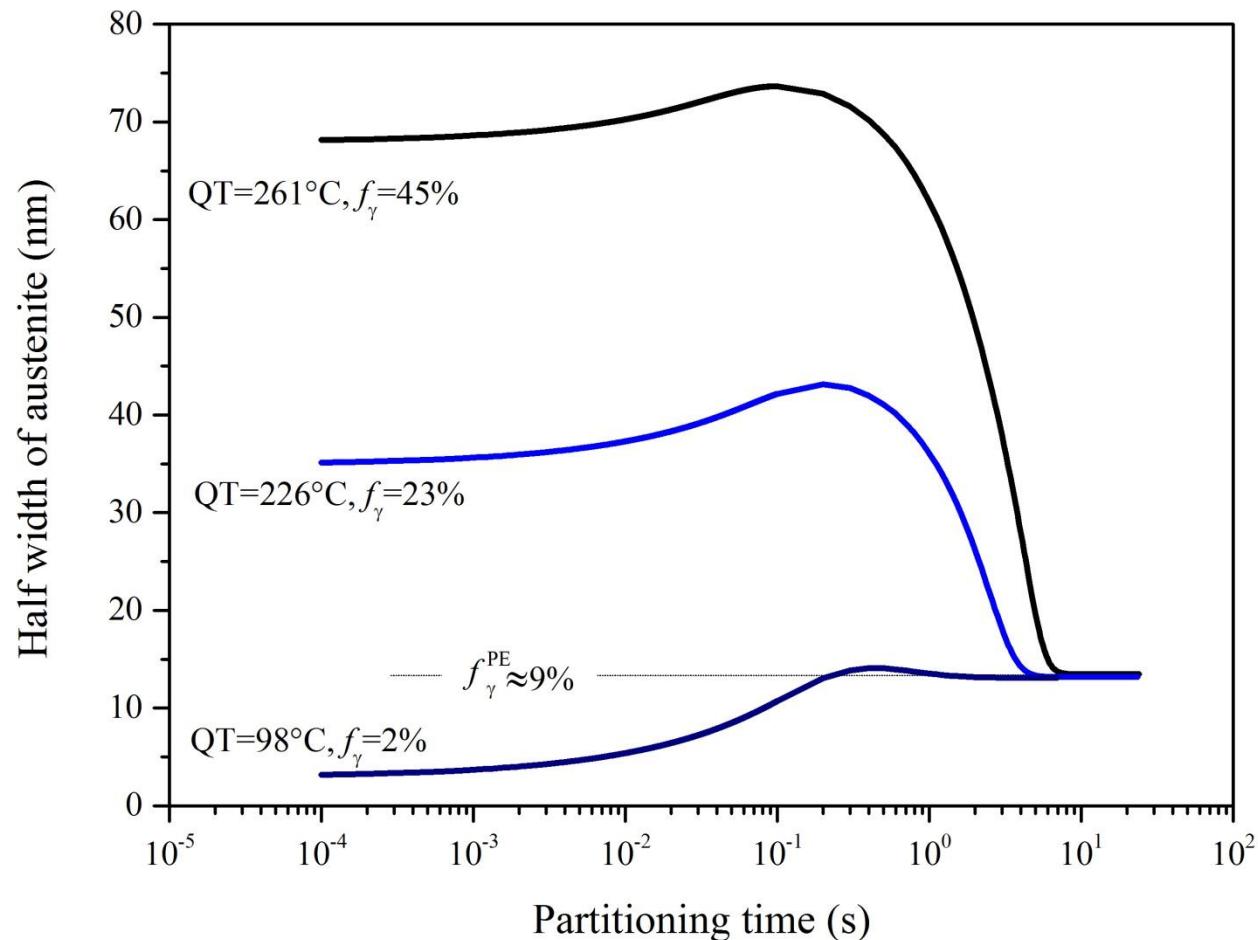
C diffusion
in α' controlled

C diffusion
in γ controlled



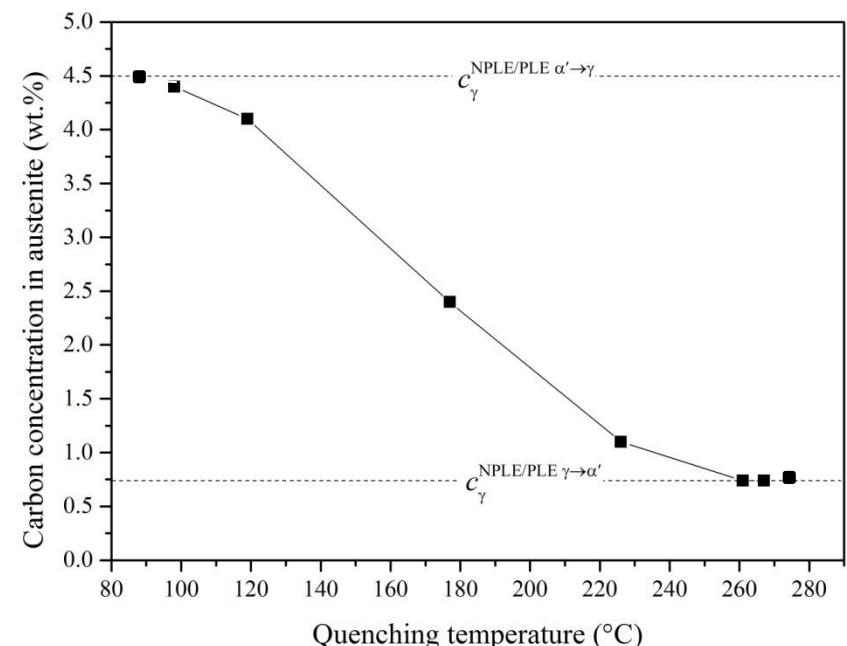
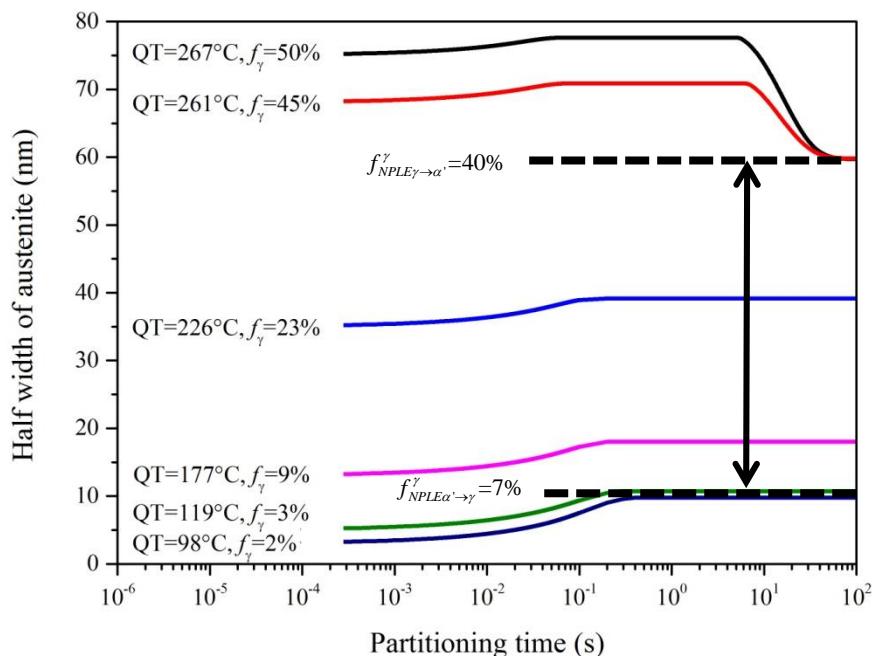
Simulations by the QP-PE Model

QT=261 / 226 / 98 °C, PT=400 °C



Simulations by the QP-LE Model

QT=267 ~ 98 °C, PT=400 °C



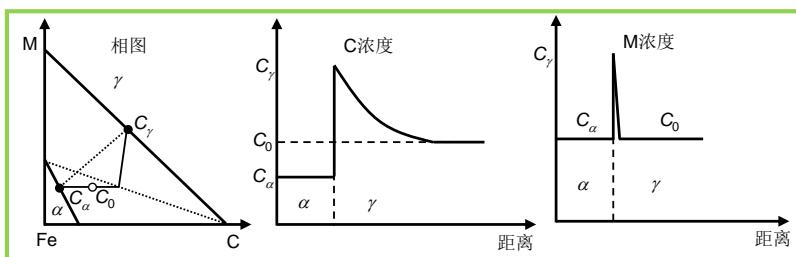
C content in austenite is between $c_\gamma^{NPLE/PLE\gamma \rightarrow \alpha'}$ and $c_\gamma^{NPLE/PLE\alpha' \rightarrow \gamma}$!

QP-LE Model

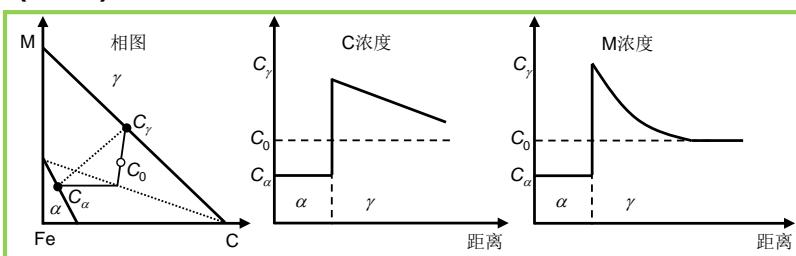
QP-LE--Local equilibrium:

$$\mu_i^\gamma = \mu_i^\alpha$$

Negligible Partitioning-Local Equilibrium (NP-LE) : C diffusion controlled



Partitioning-Local Equilibrium (P-LE) : M diffusion controlled



Diffusion Controlled Kinetic Process :

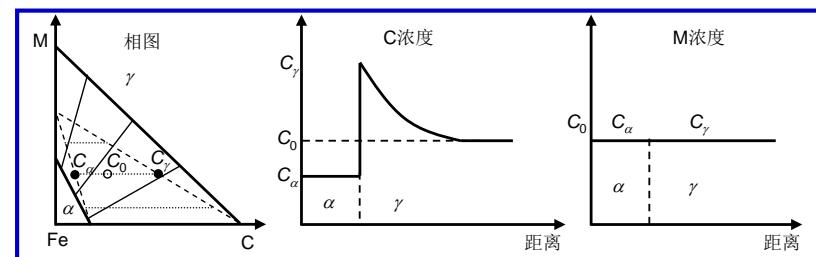
$$\frac{c_{i,j+1} - c_{i,j}}{\Delta t} = D \frac{c_{i-1,j} - 2c_{i,j} + c_{i+1,j}}{\Delta x^2} + \nu \frac{N-i}{N-1} \frac{c_{i+1,j} - c_{i-1,j}}{2\Delta x} \quad (2)$$

QP-PE--Paraequilibrium:

$$\mu_C^\gamma = \mu_C^\alpha$$

$$(\mu_M^\gamma - \mu_M^\alpha) = -\frac{X_{Fe}}{X_M} (\mu_{Fe}^\gamma - \mu_{Fe}^\alpha)$$

PE : C diffusion AND Mn indiffusible



$$\frac{\partial c}{\partial t} = \nabla \cdot (D \nabla c) \quad (1)$$

$$\nu = \left(D^\alpha \frac{\partial c^\alpha}{\partial x} - D^\gamma \frac{\partial c^\gamma}{\partial x} \right) / (c^{\gamma/\alpha} - c^{\alpha/\gamma}) \quad (3)$$