

Gas Turbine Materials/Components Life Evaluation & Extension Programs

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Presentation Outline

- Introduction to NRC Programs on GT
- GT Component Life Estimation and Extension
- Capability Guide
- Surface Modification Programs at NRC
- Some Examples of recent initiatives
- Concluding Remarks

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NRC Key Sectors

Electronic Instruments



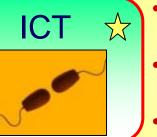




Agriculture



Criteria



Important, or will be important, to the Canadian economy

R&D and knowledge-intensive - innovation plays a key role to their success

Ones in which NRC's innovation impact can make a significant positive contribution to Canada's economy

Automotive



Construction



Chemicals



Bio-Pharma



Manufacturing



NRC-IAR structure

Director General

Aerodynamics Laboratory

Fixed-wing Aerodynamics

Rotary-wing Aerodynamics

Surface Vehicles and Structures

Montreal Road Wind Tunnel Facilities

Uplands Wind Tunnel Facilities

Aerospace Manufacturing Technology Centre

Automation & Intelligent Manufacturing

Forming & Joining of Metallic Products

Fabrication & Joining of Composite Structures

Material Removal

Flight Research Laboratory

Airborne Research

Flight Mechanics & Avionics

Flight Recorder Playback Centre

Gas Turbine Laboratory

Internal Aerodynamics & Combustion

Propulsion Performance and operability assessment

National Facilities - Icing, Altitude, Compressors and Exhausters

Lubrication Tribology

Structures & Materials Performance Laboratory

Aeroacoustics & Structural Dynamics

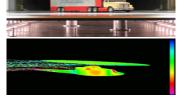
Metallic and Ceramic Materials

> Non-Destructive Evaluation

> > Structures

Composites & Novel Airframe Materials

















Compressor

mV_{aircraft}

1500

0

40





Exhaust Nozzle

mV_{jet}



•Manufacturing

Processes/Materials

- -Forging
- -DS/Single Crystal Processing
- -Fabrication/Joining (FSW,LW & EBW)
- -Compressor Coatings
- –HSM of Ti Alloys
- -Nano-materials
- -Sensing Devices MEMS
- -Hot Section Coatings
 - •EB-PVD, TBCs
 - •HVOF& Plasma Coatings
 - •CVD Coatings

•Failure Modes

- -Fatigue (LCF,HCF, Fretting & TMF)
- -Hot Corrosion
- -Cyclic Oxidation
- -Fretting/Wear
- -Erosion/Corrosion
- -Creep
- -Failure Mode interaction

Gas Turbine Engine Life Cycle

Shaft

Combustor

Turbine

•Repairs/Reworks

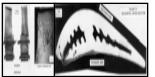
- -Single Crystal and DS Turbine Blade & Vane Repair (EBW, LW, HVOF)
- -FOD Repair/Rework
- -Fatigue Damage Repair
- -Advanced Coatings
 - •TBCs
 - •CFG coatings
 - •Erosion Resistant
- -HIP Rejuvenation
- -Diffusion Brazing

Temperature

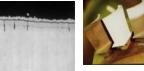


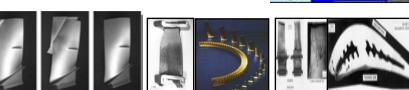
Pressure

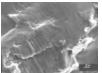








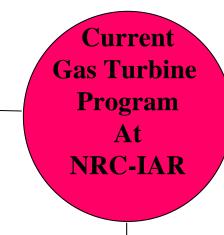






Gas Turbine Lab(GTL)

- Performance
- •DPHM
- •Internal Aerodynamics
- •Combustion
- •Instrumentation
- •Icing Certification
- •Tribolgy of Bearings
- •Weight Reduction
- **•Emission Low NOx**



Aerospace Manufacturing Technology Centre (AMTC)

- •Forming & Joining
- Automation & Robotics
- •High Speed Machining
- •Composite Structures

Structures & Materials Performance Lab(SMPL)

- •Component Design
- & Analysis(FEM),DT Analysis
- •Materials for GT
- Coatings for GT
- •GT Repair Technologies
- •Sensors & MEMS
- •Physics of Component Failure
- Coupon Testing
- •Component Testing (Vibration HCF, Spin pit, Burner rig)
- •ND Inspection
- •Modeling & Simulation(HPC)
- •Qualification and Certification

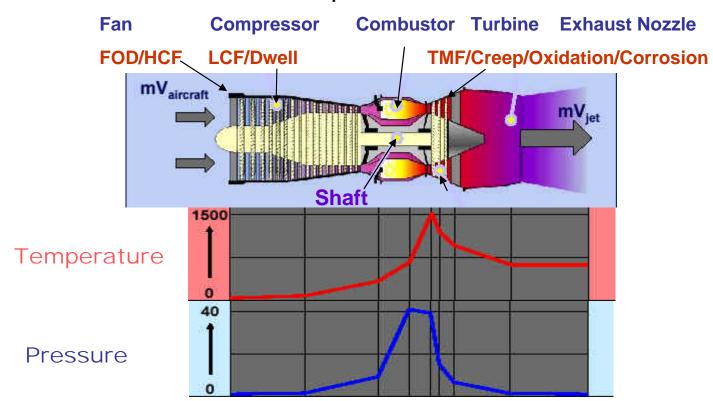
Industrialization of GTs

- •Design Change- Materials, Coatings, Surface Modifications
- •Relifing-design duty Cycle
- •Change of Failure Modes LCF/TMF to Creep
- •Hot Corrosion as Primary
 Failure Mode due to fuel derating



Physics Based Failure Mechanisms

- Understanding engine component failure mechanisms
- Develop physics-based damage accumulation models
- Model Holistic Method for life prediction





Physics of Failure & Lifing

- FOD induced HCF failure
- FEM models for vibration analysis.
- Modeling Creep fracture (for single crystal materials)
- Modeling LCF, TMF and Fretting Fatigue
- Modeling Erosion/Corrosion in the LPC
- Modeling High Temperature Erosion in the HPC
- High Temperature Oxidation Model
- High Temperature Corrosion Model for marine/industrial Environment
- Operating Mission Analysis
- Development of Life consumption rules under multiple Failure Mode conditions
- Degradation of Coating impacting bulk material properties
- Degradation of TBCs

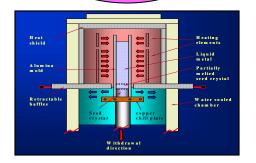
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Holistic Modeling of Gas Turbine Component Life at NRC

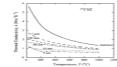
First Principle
Modeling
(Materials Design)

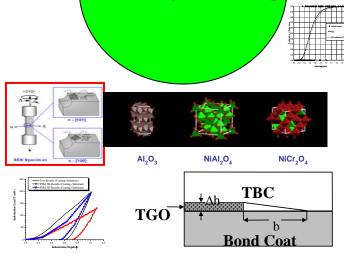
y_o y_o, interface

Process Modeling (Chemical/Mechanical)









Durability Modeling

A schematic of TBC fracture.

$$\sum_{N=1}^{N_f} (1 - \frac{\mathcal{E}_r}{\mathcal{E}_f}) (\frac{W_N}{W_c})^m + \frac{\mathcal{E}_r}{\mathcal{E}_f})^b = 1 \qquad K = \frac{Eh}{\sqrt{2 \pi b}}$$



High Temperature Materials Testing-Test Standards

Evamples:

ASTM E83

Conform to ASTM Standards with modifications to procedures as per customer requests:

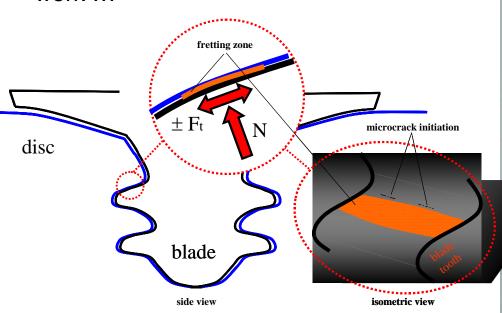
		Examples:
•	Tensile - Ambient Temperature	ASTM E8
•	Tensile - Elevated Temperature	ASTM E21
•	Stress Rupture	ASTM E139
•	Creep Strain	ASTM E139
•	Load Control Low Cycle Fatigue	ASTM E466
•	Strain Control Low Cycle Fatigue	ASTM E606
•	Crack Propagation	ASTM E647
•	Fracture Toughness (Plane-Strain)	ASTM E399
•	Axial High Cycle Fatigue	ASTM E466
•	Thermo mechanical Fatigue	ASTM E2368
•	Test Frame Alignment (Calibration)	ASTM E1012
•	Force Verification (Calibration)	ASTM E4

Verification of Extensometry

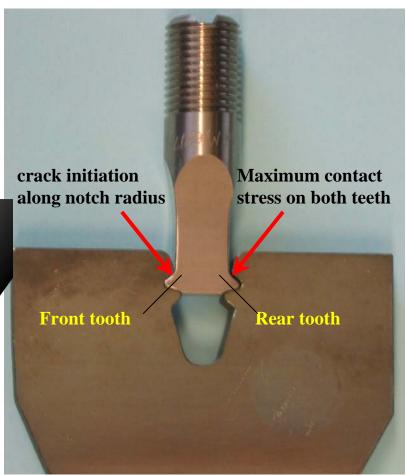
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Some examples of our work ...

Fretting/Fatigue Effects on LCF life of SX Superalloys

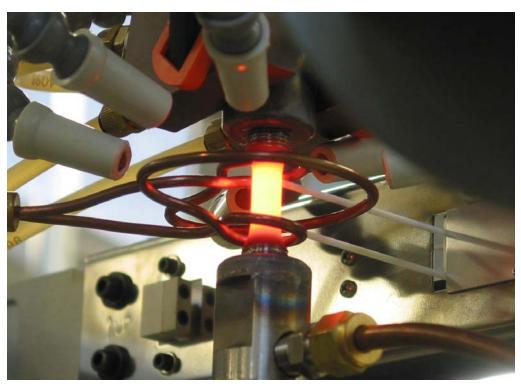


Quantification of Peening Credit on Blade Fixing by OEM

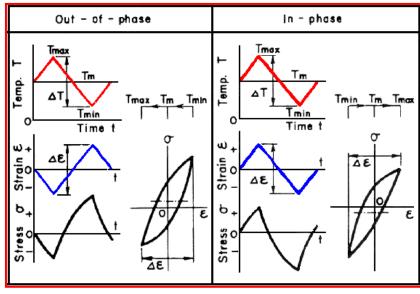




TMF Initiation Life of SX Superalloys



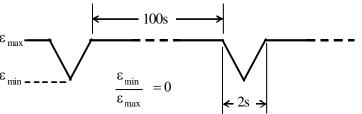
TMF Testing Programs (For Canadian & US Engine OEM's)



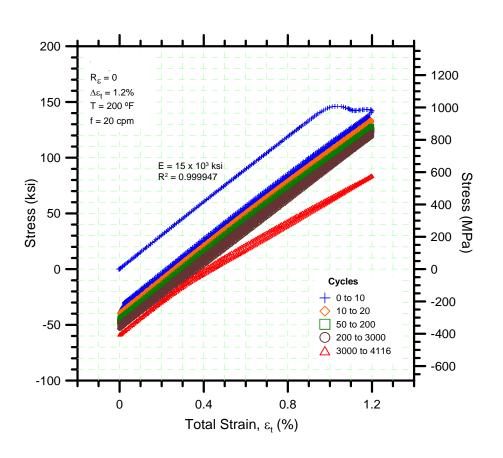
* Note: IAR has been requested to partake in TMF interlaboratory Round Robin Program







Strain Control LCF Dwell Test for a New Superalloy





NRC Capabilities in Surface Technologies

- <u>IAR-Ottawa/Montreal</u>: Process Development & Certification Testing
 - Magnetron sputtering
 - Vacuum Arc
 - EB-PVD
 - Shot Peening
 - Laser Alloying
- IMI- Montreal: Thermal spraying
 - Plasma, HVOF, arc, flame, cold spray
 - Suspension/solution plasma spraying
- IMI-London: Laser processing
 - Pulse laser deposition
 - Laser cladding, laser transformation hardening, laser glazing, laser alloying, laser gas nitriding
 - Laser-assisted plasma spraying
- IFCI-Vancouver: Wet processing:
 - Chemical surface modification, sol gel, spray pyrolysis,
 - Reactive spray deposition technology

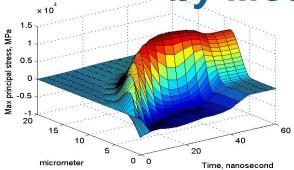


NRC Capabilities in Surface Technologies

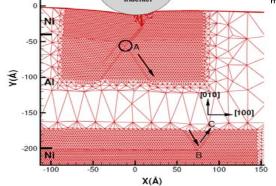
- Modeling & Design
 - Coating Design
 - Coating Process simulation
 - Coating durability simulation
- Testing & Evaluation
 - Erosion, abrasion, hardness, cyclic oxidation, fretting
 - Thermal properties, mechanical properties
 - Burner rig
 - Engine
- Characterization, non-destructive evaluation
 - SEM, AFM, X-ray diffraction, Nano Hardness
 - Laser-ultrasonics
 - Neutron reflectometry

NRC-CNRC

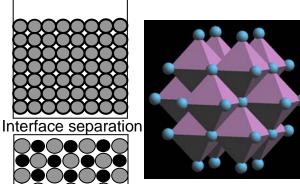
Coating Design by Modeling



Coating-particle interactions by dynamic FEM modeling



Coating indentation deformation by MD/FEM simulation



Materials properties prediction by first-principle DFT calculation



Oxidation/Corrosion Testing

Internally cooled TBC pins

Solid Pt-Al coated pins

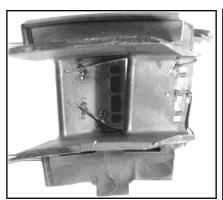
Hot corrosion with retort

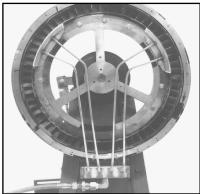


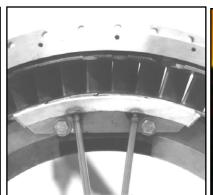




T56 1st stage nozzle guide vanes







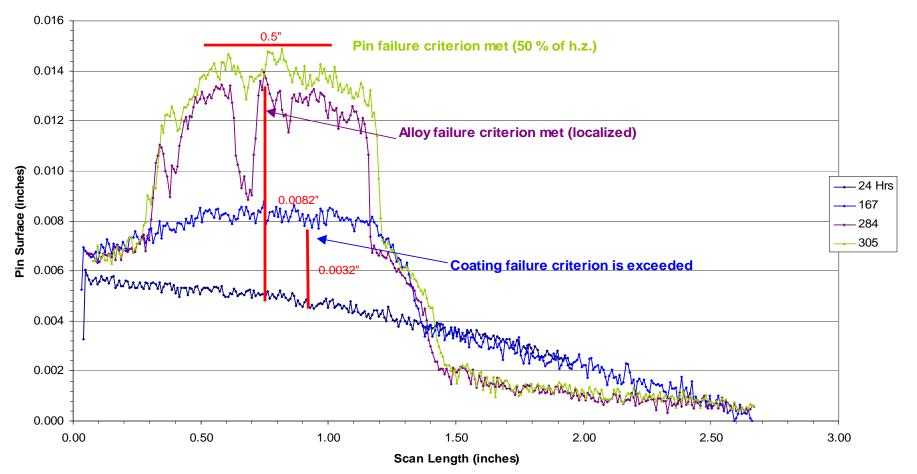


Burner Rig Test Facilities

NRCaerospace.com

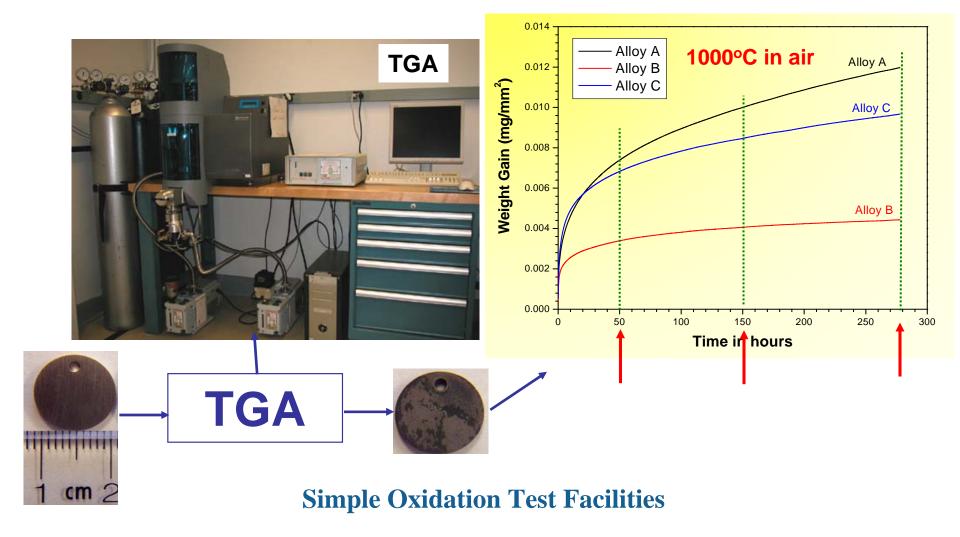
Damage Progression in Substrate & Coatings

Laser measurements quantifying coating loss





Oxidation/Corrosion Testing





Magnetron Sputtering & Cathodic Arc Coating Processes for Gas Turbine Component Life Extension



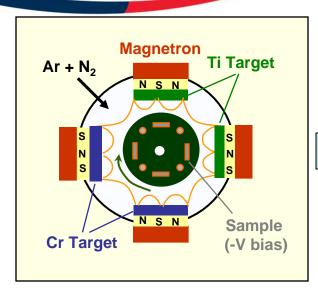


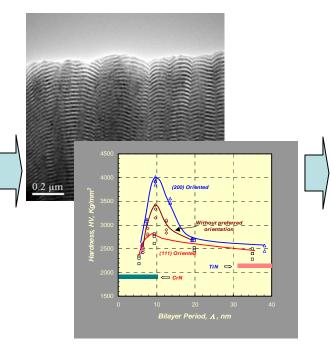
Magnetron Sputtering PVD

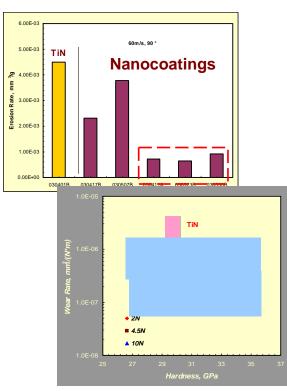
Cathodic Arc PVD

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Nanostructured Superhard Coatings







PVD Processing

Plasma etching
Substrate bias
Working pressure
Temperature
Fixturing

Properties:

Thickness: 0.5 – 10 µm Hardness: 30 – 40 GPa Good thermal stability Low coefficient of friction High adhesion strength

Performance:

Excellent wear resistance Superior erosion protection



High Temperature Corrosion in Industrial/Marine Turbines

Hot Corrosion

Type I hot corrosion —high temperature reaction between a coating, a molten deposit of alkaline salts, and sulfur-containing combustion gas. It includes basic fluxing, acidic fluxing, and alloy-induced acidic fluxing.

Type II hot corrosion — low temperature reaction between a coating, a solid deposit of alkaline salts, and sulfur-containing combustion gas. This corrosion produces low-point-eutectic phases of Ni and Co which become molten at 650-800°C.



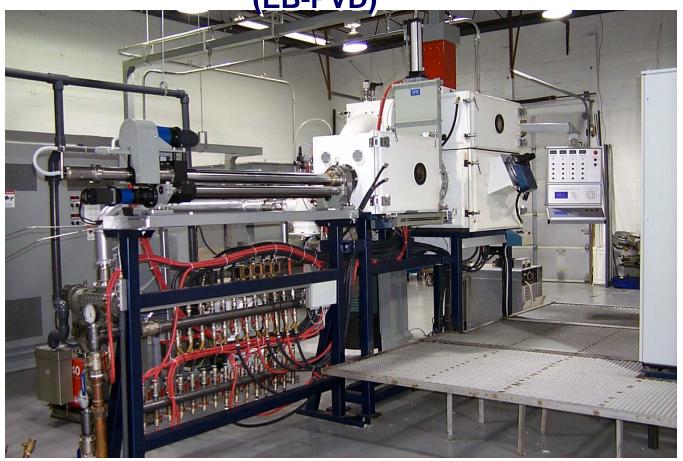
Combating Hot Corrosion

For Industrial/Marine Gas Turbines:

- Diffusion Coatings
 - Si-modified slurry coating
 - Pt-modified CVD coating
 - Cr-modified pack coating
- Overlay Coatings
 - » MCrAIY by EB-PVD
 - » MCrAIY by APS, LPPS, and VPS
 - » MCrAIY by HVOF
- Thermal Barrier Coatings (TBC)
 - » YSZ + MCrAIY bond coating by EB-PVD

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High-Rate Electron-beam Physical Vapor Deposition (EB-PVD)



Multipurpose NTI High-Rate EB-PVD Coater with reactive plasma-assisted deposition capability



NRC-NTI Aerospace Coatings Development Center (NRC-NTI-ACDC)

National Research Council of Canada – Institute for Aerospace Research

Structures and Materials Performance Laboratory

Gas Turbine Laboratory



NRC-NTI Aerospace Coatings Development Centre (NRC-NTI-ACDC)



Newman Technologies Inc.



Objectives of NRC-NTI-ACDC

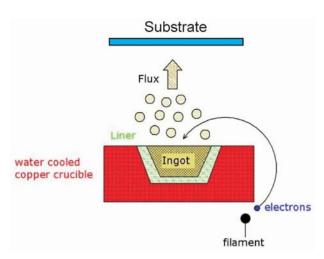
- Design & fabrication of novel Gas Turbine materials.
- Design & Fabrication of advanced Gas Turbine protective coatings.
- Design of new PVD-based coating processes.
- Design of customized PVD-based coating equipment.
- Ideal tool for addressing Coatings for Marine Applications
- Assistance in Technology Transfers to Clients & Partners.



Comparison of Traditional and High-Rate EB-PVD

Traditional EB-PVD

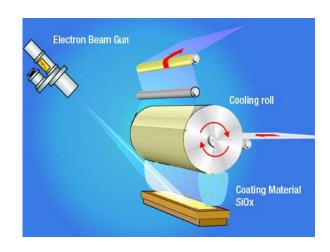
(e-gun power is below 30 kW, deposition rate below 0.1 μm/min)





High-Rate EB-PVD

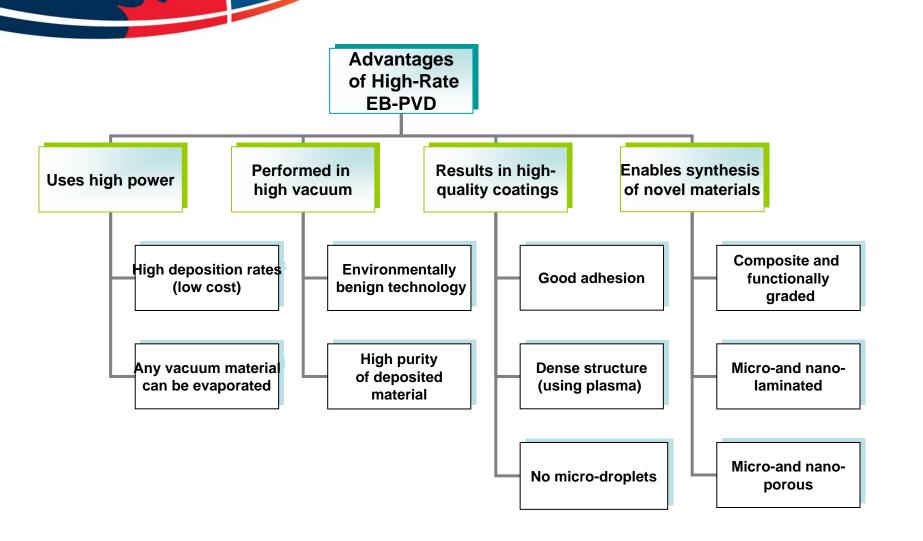
(e-gun power exceeds 60 kW, deposition rate exceeds 10 µm/min)





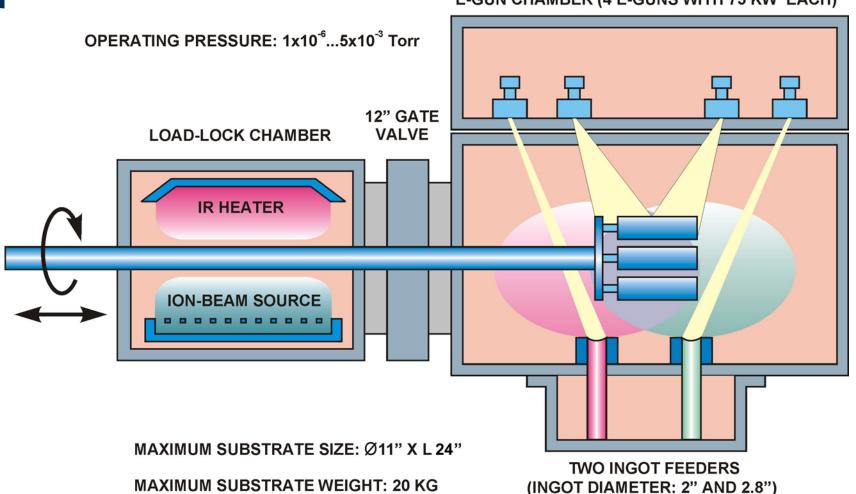


Advantages of High-Rate Deposition EB-PVD Technology



General Design of The EB-PVD Coater

E-GUN CHAMBER (4 E-GUNS WITH 75 KW EACH)







EB-PVD without plasma assistance

Synthesis of Composite and Graded Materials by Reactive Plasma-Assisted High-Rate EB-PVD



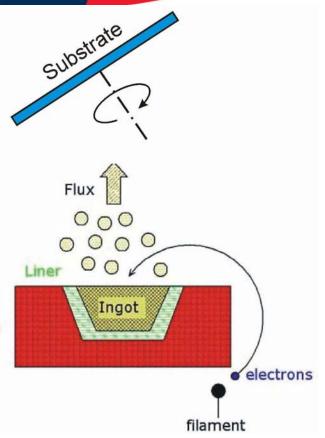
EB-PVD with plasma assistance

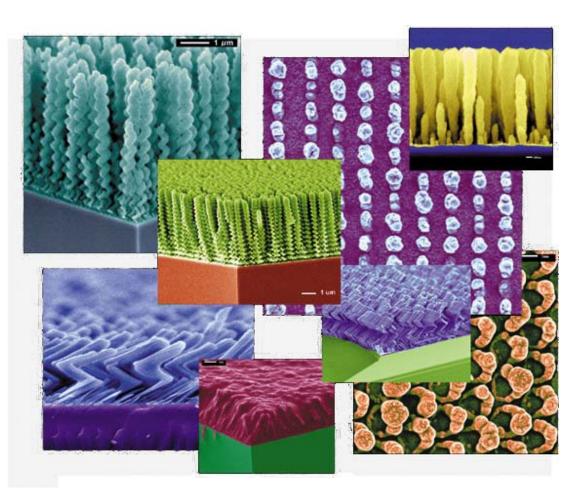
The effect of plasma assistance:

- densifies coating microstructure;
- improves coating adhesion;
- stimulates chemical reactions;
- enables control of intrinsic stress.



Glancing Angle Deposition of Materials with Highly Developed Morphology

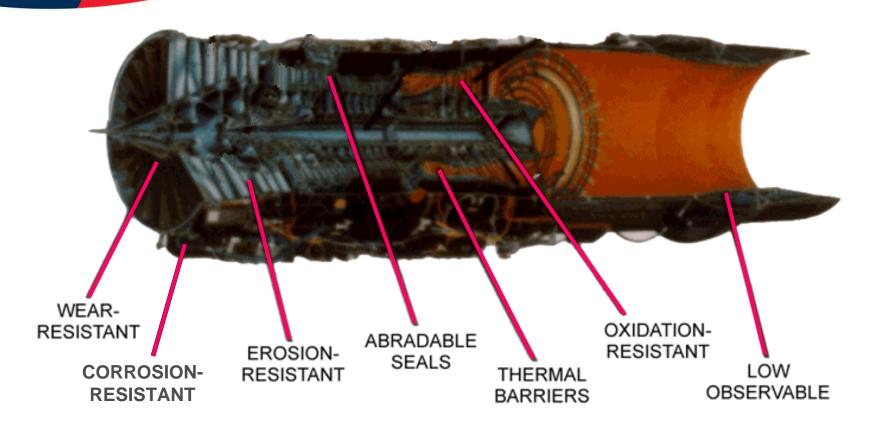




Examples of sculptured coatings with helicoidal and nematic microstructure



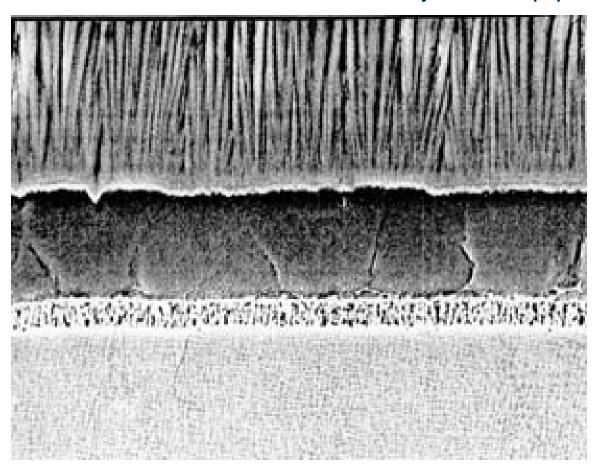
Fabrication of Advanced Gas Turbine Protective Coatings



NRC-CNRC NRCaerospace.com

Fabrication of Duplex Thermal Barrier Coatings with Improved Properties

Duplex (Two-Layer) Thermal Barrier Coating fabricated by two-step process



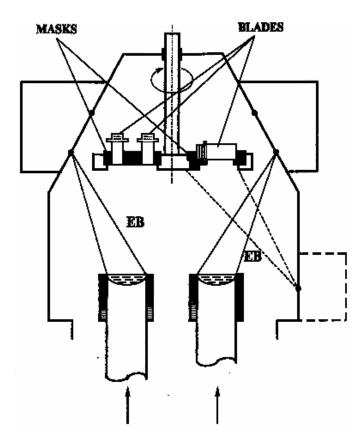
YSZ ceramic top
coat applied by
EB-PVD

MCrAIY bond coat
applied by thermal spray
or EB-PVD

Substrate

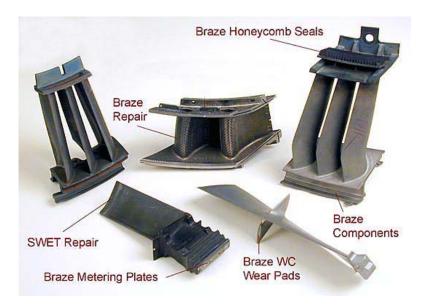


Typical arrangement for blade repair



Dimensional Restoration of Turbine Blades Using EB-PVD

- Blade tips can be repaired by positioning the blades vertically with tips facing the vapour stream
- Airfoils can be repaired by positioning the blades horizontally





Airborne sand turns into Calcium-Magnesium-Alumino-Silicate when reacts with TBCs

- CMAS attacks the TBC and degrades it
- CMAS impregnated TBC has low strain-tolerance and highly susceptible to TMF loading
- Addition of Alumina and Rutile to TBC has resulted some success but alumina raises the conductivity of the TBC

Degradation of TBCs by CMAS Attack

A. Aygun et al. | Acta Materialia 55 (2007) 6734-6745

CMAS Glass

HeatTreatment

ZrO₂ with Y²·*₄AP²·*₇Ti⁴· Solute
ZrO₂ depleted in Y²·, AP³·, Ti⁴· Solute

Glass Comp. in
Psuedowollastonite Field

Glass Comp. in

Fig. 14. Schematic illustration (not to scale) of the proposed mechanism of CMAS-attack mitigation in SPPS(YSZ+Al+Ti) TBCs

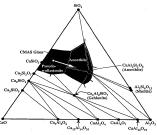


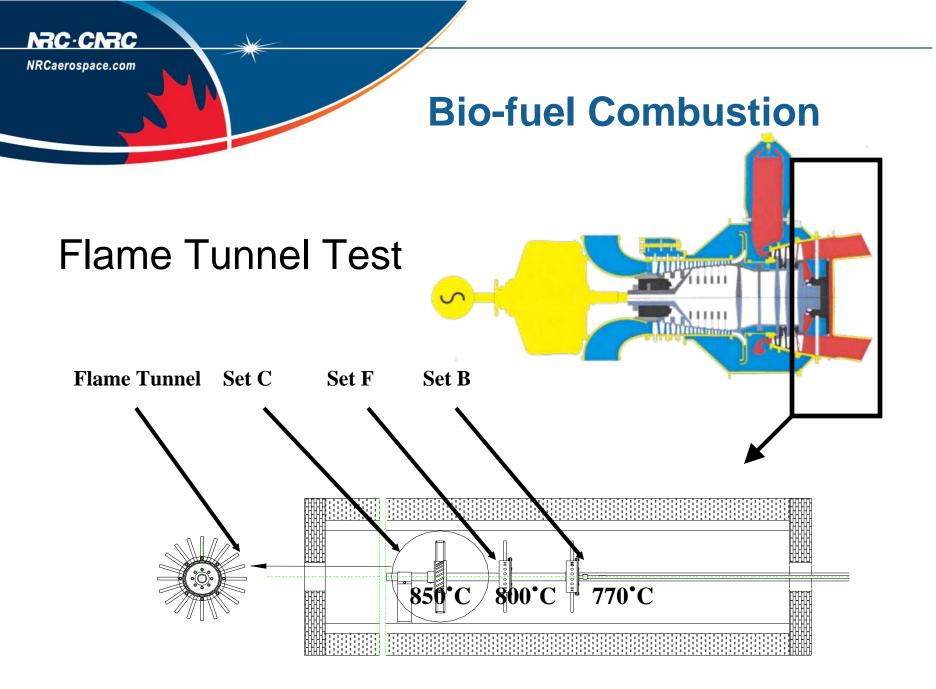
Fig. 15. Redrawn CaO-Al₂O₃-SiO₂ (wt.%) ternary phase diagram (after [28]) showing composition of the simulated CMAS glass, and the psuedowollastonite and anorthite fields. The arrow indicates composition shift in the CMAS glass.

Hutchinson [37] analyzed the thermomechanical failure of CMAS-penetrated TBC in a thermal gradient, and indicated that as the CMAS-penetration depth increases the propensity of thermomechanical failure of the top-cat increases in a complex way. A critical CMAS-penetration depth could be defined and compared with the activation CMAS-penetration depth of 60 μm. However, that critical CMAS-penetration depth would depend on various factors, including thermal gradients, cooling transients, surface temperature, and thermal and mechanical properties of the TBC constituents [37], many of which are unknown for the present system. Thus, consideration of these issues is beyond the scope of the present paper.

Finally, the SPPS process is very versatile in being able to incorporate metastably a wide range of solutes in a variety of oxide coatings. Thus, the engineered TBCs can serve as reservoirs of tailored solutes that can interact with not only CMAS deposits, but also other types of deposits (salts, ash, and contaminants), and mitigate their attack

5. Summary

A new approach for mitigating CMAS attack on TBCs is presented, where up to 20 mol.% Al₂O₃ and 5 mol.% TO₂, in the form of a solid solution and not as discrete second phases, have been incorporated into YSZ TBCs. The SPPS process, which is uniquely suited for producing coatings of metastable ceramics with extended solid-solubilities, has been used to deposit such TBCs with engineered chemistries. Here, the TBC serves as a reservoir of Al and the Ti solutes, which are incorporated into the molten CMAS glass that is in contact with the TBC. An accumulation of Al concentration in the CMAS glass, as it penetrates the TBC, shifts the glass composition from the difficult-to-crystallize psuedo-wollastonite field. The incorporation of Ti in the glass





Wood Waste Derived Pyrolysis Oil

Parameter	Units	Bio-Fuel	Diesel	
Net Calorific Value	MJ/kg	15-20	42.0	
Viscosity	cSt	3-15@80 °C	2-4@20 °C	
Acidity	рН	2.3-3.0	5	
Water	wt %	19-24	0.05 v% (combined)	
Particulate	wt %	0.1-0.6		
Ash	wt %	0.01-0.5	0.01	
Alkali (Na + K)	ppm	5-400	<1	

Experimental

Materials for Hot Corrosion Test

The List of Coatings for Gas Turbine Components

Specimens	Substrate alloys	Coatings	Coating Process	Test temperature (°C)
Combustion liner	Alloy I	DC I	Si-modified slurry aluminide coating	850
		OC I	HVOF	
		DC VI	Cr-Al pack coating	
		TBC I	EB-PVD	
		TBC III	Al ₂ O ₃ +SiO ₂ barrier coating	

List of Coatings for Gas Turbine Components

Specimens	Substrate alloys	Coatings	Coating Process	Test temperature (°C)
Stage 1 and	Alloy II	DC I	Si-modified slurry aluminide coating	800
stage 2 nozzle		DC II	Si-modified slurry aluminide coating	
		DC IV	Cr-Al pack coating	
		DC V	Cr-Al pack coating	
		DC VI	PVD	
Stage 1 blade	Alloy III	DC I	Si-modified slurry aluminide coating	770
		DC III	Pt-modified CVD	
		DC IV	Cr-Al pack coating	
		TBC II	EB-PVD	
		DC VI	PVD	



Experimental

Flame Tunnel Test





Experimental

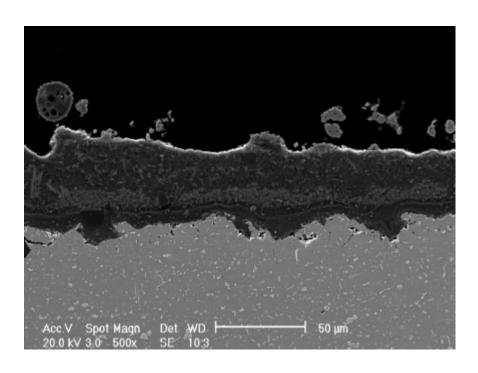
Flame Tunnel Test (continued)

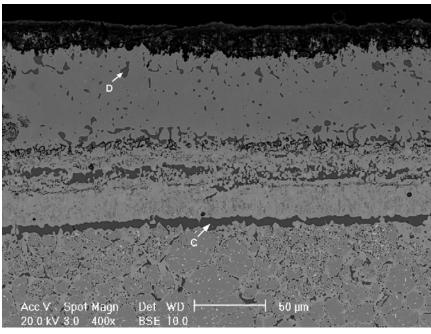




Results

Specimen Set C Specimens C42 (TBC III) Specimens C22 (DC IV)





Results

Comparison of coatings at different service temperatures

Coatings	Service Temperatures		
	850°C	800°C	770°C
DCI		++	++
OC I	++	0	0
TBC III	++	0	0
DC III	0	0	++
DC IV			++
TBC I	+	0	0
TBC II	0	0	+
DC V	0		0
DC II	0		0
DC VI	0		

^{+ +:} the coating performs very well; +: the bond coating performs well, but TBC spalled off; - -: the coating was severely corroded; o: the coating was not tested at the temperature.









- The high operating pressures of present and future engines can raise the dew point for sodium sulfate deposition to 1000°C and higher
- It is important to understand the effects of molten salt deposits on ceramic parts at high temperatures
- Past work has shown that SiC is etched by molten salts



- The hot corrosion of sintered α -SiC by thin films of Na₂SO₄ and Na₂CO₃ was studied at 1000°C in controlled gas atmospheres
- In all cases the protective SiO₂ layer dissolved to form silicate, leading to corrosion
- In all cases the presence of liquid films is responsible for rapid transport rates and the subsequent enhanced gas permeabilities.
- The reaction of sintered α -SiC with Na₂SO₄ and Na₂CO₃ produces large amounts of SiO₂ and causes dramatic etching of the SiC substrate.



- Na₂SO₄/0.01% SO₃ melts lead to rough friable scales which cause uniform pitting of the substrate
- Na₂CO₃/0.1% CO₂ and Na₂SO₄/air melts lead to smooth, glassy scales which cause uneven pitting and grain-boundary attack.
- In all cases, it appears that attack occurs by dissolution of the protective oxide film
- It appears that carbon promotes SiO₂ dissolution by formation of CO and CO₂.

Concluding Remarks

- NRC has a number of R&D programs related to Gas Turbine Operation, Durability and life extension in collaboration with their clients and partners.
 Some programs related to life extension have been presented here.
- NRC has been very active in the surface engineering sector across the council. The aerospace sector has been addressed by the IAR on PVD, IMI on Thermal Spray and Laser Technologies, respectively.
- The current challenges facing the aerospace coating industry in terms of of developing coatings for alternate fuel usage in the Gas Turbines have been identified and NRC collectively is ready to respond to the industry needs.
- The collaboration with NTI in terms of EB-PVD coating technologies brings in an ideal and timely opportunity for technology collaboration, joint development, Technology Demonstration and commercialization.



Questions?



